

Short Communication

Effect of Nitro Groups on Electrochemical and Photovoltaic Behavior of Tris-thiophenes used in Organic Solar Cells.

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4-[3⁵-(4-aminophenyl)[1²,2²:2⁵,3²-terthiophen]-1⁵-yl]benzotrile (CB4TA), 4-[3⁵-(4-aminophenyl)[1²,2²:2⁵,3²-terthiophen]-1⁵-yl]-2-nitrobenzotrile (C2B4TA) and 4-[3⁵-(4-aminophenyl)[1²,2²:2⁵,3²-terthiophen]-1⁵-yl]-2,6-dinitrobenzotrile (C2,5B4TA), the first molecule without nitrogroups and the others with electron withdrawing nitro-groups, were successfully synthesized. This family of compounds has acceptor and donor moieties in the same molecule. The acceptor and donor functions are separated by a “molecular bridge”. All molecules were studied to evaluate the effect of nitro groups on the photovoltaic yield in an organic solar cell (OPVC). CB4TA and the related molecules were characterized via NMR-spectroscopy, optical and electrochemical methods and then deposited by high vacuum sublimation over ITO treated with MoO₃ and CuI as buffer layers. Subsequently the morphology of deposits was characterized using atomic force microscopy (AFM) and X-ray diffraction (XRD). The photovoltaic devices were built under the same conditions, observing differences in the photovoltaic yield. Results show that the nitro groups have important effects over frontier orbitals and photovoltaic yield.

Keywords: Electron withdrawing effect, nitro-groups, morphology, XRD, electrochemistry, photovoltaic yield.

1. INTRODUCTION

Molecules with donor and acceptor groups separated by a conducting conjugated π -system have gained increasing interest as they have at the same time high stability, electron-conductivity and their optical properties can be fine-tuned. Several members of this family have shown good electrical yields in photovoltaic devices. [1-4].

Donor acceptor molecules have attracted particular attention, due to outstanding environmental stability, low band gap value, good conductivity, charge transfer and reasonable photovoltaic yield.[5-9] For this reason, donor- π -bridge-acceptor molecules are materials with photoelectrical properties that could be used in OPVC. To increase efficiency in technological development and succeed in the design of new and improved materials it is important to find out the relationship between photoelectrical properties, electronic structure morphology and the electron-withdrawing (in our case nitro groups) effect of functional groups on these devices. With respect to the morphology it is very important to establish a good contact between the donor-acceptor molecule and fullerene which is used as standard molecule to accept electrons in multi heterojunction photovoltaic devices. [10]

The energy gap (E_g) between the highest occupied molecular orbital (HOMO) and the energy of the lowest unoccupied molecular orbital (LUMO) is an important parameter to decide if a molecule can be used in a photovoltaic device as well as to understand and to tune the optical, electrical and chemical properties of these molecules by chemical modifications. This parameter can be estimated from electrochemical and optical (by UV-vis spectroscopy) measurements [11]

In this work we report how the introduction of nitro-groups in the molecules (CB4TA), (C2B4TA), (C2,5B4TA) can change their behavior in OPVCs. It causes higher light absorption, promotes both better inter- and intra-molecular charge transfer (determined by J/V curves and photovoltaic yield) and produces better π - π stacking/aggregation (by XRD) leading to a small energy band gap [12].

2. EXPERIMENTAL DETAILS

All chemicals and solvents were purchased from *Aldrich*, and were used as received. All products obtained were characterized using NMR and UV-vis spectroscopy. Structures were confirmed by NMR spectra recorded on a *Bruker* 400 MHz spectrophotometer using DMSO d_6 as solvent.

Molecular structures are shown in figure 1

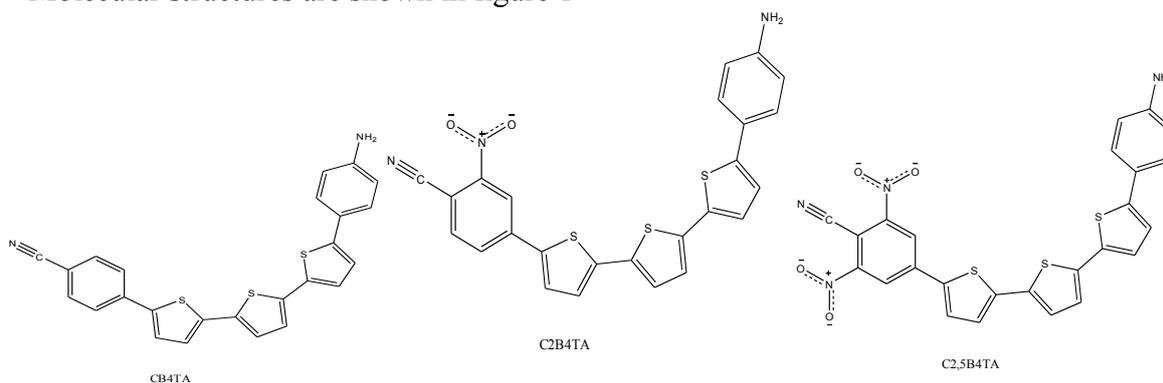
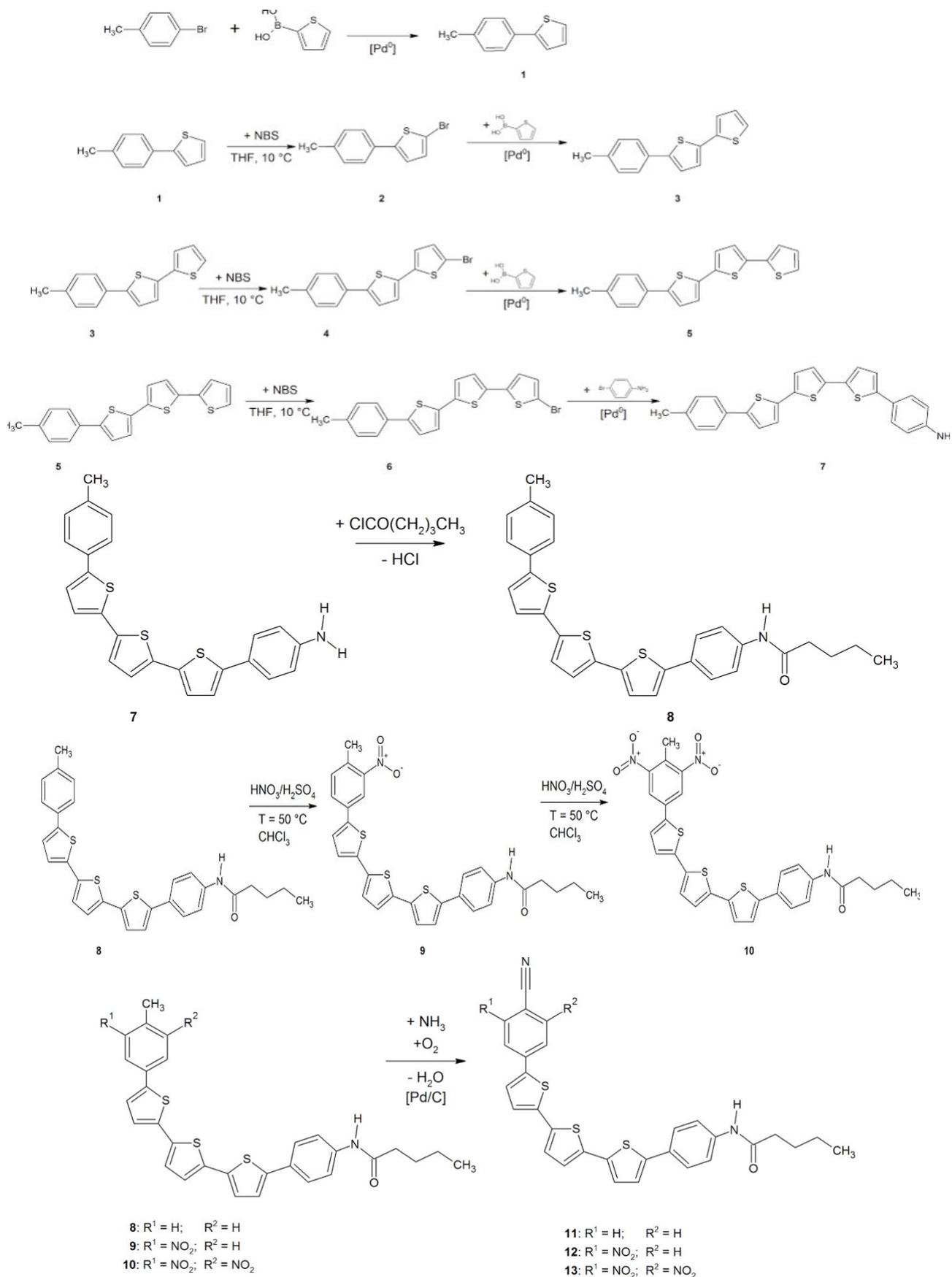
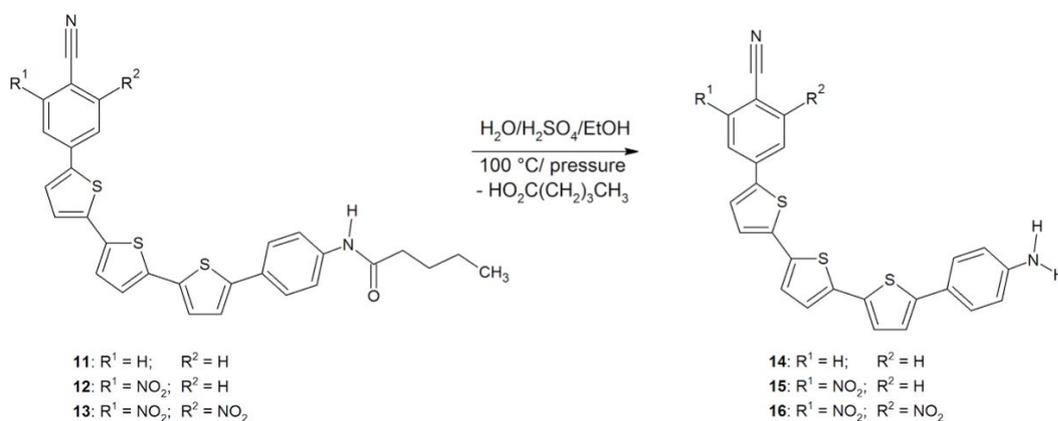


Figure 1. Structures of molecules used as electron donors in OPVC

The synthesis of molecules was made as follow:





Synthetic Procedure:

Synthesis of 4-methyl-2-(2-thienyl)benzene 1

2.5g of p-bromotoluene and 3.5g of 2-thienylboronic acid were heated in 200 mL of anhydrous THF in presence of 10% Pd/C as catalyst for 48 h in a sealed tube under argon atmosphere in a sealed high pressure tube. Once cooled down to room temperature the reaction mixture was transferred to a separation funnel equipped with 30mL of saturated aqueous NaCl-solution and extracted 4 times with 50mL ethyl acetate EtOAc each. The organic phase was dried over MgSO₄, filtered and concentrated on a rotary evaporator. The residue was re-dissolved in 50mL of EtOAc and purified via column chromatography on silica gel using EtOAc/petrol ether 10:1 as eluent.

Purity of the fractions was monitored by TLC. Pure fractions were united and evaporated under reduced pressure yielding 2g (77%) of **1**. The resulting product characterized by ¹H NMR spectroscopy.

Synthesis of 2

2g of **1** and 500mg of NBS were dissolved in 200mL of THF and stirred for 3 h at 10°C. Then the reaction mixture was transferred to a separation funnel equipped with 30mL of saturated aqueous NaCl-solution and extracted 4 times with 50mL ethyl acetate EtOAc each. The organic phase was dried over MgSO₄ and evaporated under reduced pressure. The residue was redissolved in 50mL of EtOAc and the product was purified by column chromatography on silica gel using EtOAc/n-hexane 5:1 as eluent.

Purity of the fractions was monitored by TLC. Pure fractions were united and evaporated under reduced pressure yielding 1.5g (75%) of **2**. The resulting product was characterized by ¹H NMR spectroscopy.

Synthesis of 3

1.5g of **2** and 2.1g of 2-thienylboronic acid were heated in 200mL anhydrous THF for 48h in presence of 10% Pd/C as catalyst in a sealed tube under argon atmosphere. Once cooled down to room temperature the reaction mixture was transferred to a separation funnel equipped with 30mL of saturated aqueous NaCl-solution and extracted 4 times with 50mL ethyl acetate EtOAc each. The

organic phase was dried over MgSO₄, filtered and concentrated on a rotary evaporator. The residue was re-dissolved in 50mL of EtOAc and purified via column chromatography on silica gel using EtOAc/n-hexane 7:1 as eluent.

Purity of the fractions was monitored by TLC. Pure fractions were united and evaporated under reduced pressure yielding 1.4g (93%) of **3**. The resulting product was characterized by ¹H NMR spectroscopy.

Synthesis of 4

1.4g of **3** and 467 mg of NBS were dissolved in 200 mL of THF and stirred for 3 h at 10°C. Then the reaction mixture was transferred to a separation funnel equipped with 30mL of saturated aqueous NaCl-solution and extracted 4 times with 50mL ethyl acetate EtOAc each. The organic phase was dried over MgSO₄ and evaporated under reduced pressure. The residue was redissolved in 50mL of EtOAc and the product was purified by column chromatography on silica gel using EtOAc/n-hexane 5:2 as eluent.

Purity of the fractions was monitored by TLC. Pure fractions were united and evaporated under reduced pressure yielding 1.2g (88 %) of **4**. The resulting product was characterized by ¹H NMR spectroscopy.

Synthesis of 5

1.2g mg of **4** and 1.7g of 2-thienylboronic acid were heated in 100mL anhydrous THF for 48h in presence of 10% Pd/C as catalyst in a sealed tube under argon atmosphere. Once cooled down to room temperature the reaction mixture was transferred to a separation funnel equipped with 30mL of saturated aqueous NaCl-solution and extracted 4 times with 50mL ethyl acetate EtOAc each. The organic phase was dried over MgSO₄, filtered and concentrated on a rotary evaporator. The residue was re-dissolved in 50mL of EtOAc and purified via column chromatography on silica gel using EtOAc/n-hexane 8:3 as eluent.

Purity of the fractions was monitored by TLC. Pure fractions were united and evaporated under reduced pressure yielding 1g (83 %) of **5**. The resulting product was characterized by ¹H NMR spectroscopy.

Synthesis of 6

1g of **5** and 400mg of NBS were dissolved in 200 mL of THF and stirred for h 3 h at 10°C. Then the reaction mixture was transferred to a separation funnel equipped with 30mL of saturated aqueous NaCl-solution and extracted 4 times with 50mL ethyl acetate EtOAc each. The organic phase was dried over MgSO₄ and evaporated under reduced pressure. The residue was redissolved in 50mL of EtOAc and the product was purified by column chromatography on silica gel using EtOAc/n-hexane 5:1 as eluent.

Purity of the fractions was monitored by TLC. Pure fractions were united and evaporated under reduced pressure yielding 880 mg (88%) of **6**. The resulting product was characterized by ¹H NMR spectroscopy.

Synthesis of 7

880 mg of **6** and 1.2g of 4-aminophenyleboronic acid were heated in 100 mL of THF in presence of 10% Pd/C as catalyst for 48h in a sealed tube under argon atmosphere. Once cooled down to room temperature the reaction mixture was transferred to a separation funnel equipped with 30mL of saturated aqueous NaCl-solution and extracted 4 times with 50mL ethyl acetate EtOAc each. The organic phase was dried over MgSO₄, filtered and concentrated on a rotary evaporator. The residue was re-dissolved in 50mL of EtOAc and purified via column chromatography on silica gel using EtOAc/n-hexane 8:3 as eluent.

Purity of the fractions was monitored by TLC. Pure fractions were united and evaporated under reduced pressure yielding 1.2g (29%) of **7**. The resulting product was characterized by ¹H NMR spectroscopy.

Synthesis of 8

1.2g of **7** were dissolved in 200 ml of anhydrous THF and 600 mL of pentanoic acid chloride were added. The mixture was heated for 40 min to 50 °C. Once cooled down to room temperature the reaction mixture was transferred to a separation funnel equipped with 30mL of saturated aqueous NaCl-solution and extracted 4 times with 50mL ethyl acetate EtOAc each. The organic phase was dried over MgSO₄, filtered and concentrated on a rotary evaporator. The residue was re-dissolved in 50mL of EtOAc and purified via column chromatography on silica gel using EtOAc/n-hexane 9:2 as eluent. Purity of the fractions was monitored by TLC. All pure fractions were united and the solvent evaporated under reduced pressure yielding 1.1g (95%) of **8**. The product was characterized by ¹H-NMR spectroscopy.

Synthesis of nitrated product 9

20 mg of **8** were dissolved in 100 mL of CHCl₃. 5 mL of conc. HNO₃ and 15 mL of conc. H₂SO₄ were added and the mixture was heated in a sealed tube under argon atmosphere for 24 h to 50 °C. Once cooled down to room temperature the reaction mixture was transferred to a separation funnel equipped with 30mL of saturated aqueous NaCl-solution and extracted 4 times with 50mL ethyl acetate EtOAc each. The organic phase was dried over MgSO₄, filtered and concentrated on a rotary evaporator. The residue was re-dissolved in 50mL of EtOAc and purified by column chromatography on silica gel with EtOAc/n-hexane 10:1 as eluent. Purity of the fractions was monitored by TLC and the desired pure fractions evaporated together yielding 22 mg (78 %) of **9**. The product was characterized by NMR spectroscopy.

Synthesis of double nitrated product 10

10 mg of **9** were dissolved in 100 mL of CHCl₃. 5 mL of conc. HNO₃ and 15 mL of conc. H₂SO₄ were added and the mixture was heated in a sealed tube under argon atmosphere for 48 h to 100 °C. Once cooled down to room temperature the reaction mixture was transferred to a separation funnel equipped with 30mL of saturated aqueous NaCl-solution and extracted 4 times with 50mL ethyl acetate EtOAc each. The organic phase was dried over MgSO₄, filtered and concentrated on a rotary evaporator. The residue was re-dissolved in 50mL of EtOAc and purified by column chromatography

on silica gel with EtOAc/n-hexane 10:1 as eluent. Purity of the fractions was monitored by TLC and the desired pure fractions evaporated together yielding 5 mg (29 %) of **10**. The product was characterized by NMR spectroscopy.

Synthesis of nitriles 11, 12 and 13

28 mg of **8** respectively 12 mg of **9** or 5 mg of **10** were dissolved in 100 mL of anhydrous ethanol. 10% Pd/C were added, and the mixture heated with NH₃ and O₂ (NH₃ and O₂ anhydrous were purchased from sigma-aldrich) in the high pressure tube for 24h to 200 °C. Then the solvents and the excess of O₂ and NH₃ were evaporated. Once cooled down to room temperature the reaction mixture was transferred to a separation funnel equipped with 30mL of saturated aqueous NaCl-solution and extracted 4 times with 50mL ethyl acetate EtOAc each. The organic phase was dried over MgSO₄, filtered and concentrated on a rotary evaporator. The residue was re-dissolved in 50mL of EtOAc, and the products purified by column chromatography on silica gel using as eluent CHCl₃/n-hexane 5:1 for **11**, EtOAc/n-hexane 4:1 for **12** and EtOAc/n-hexane 10:3 for **13**. The resulting pure fractions containing **11**, **12** and **13** respectively were evaporated and the resulting products yielding 15mg (54%) for **11**, 5mg (42%) for **12**, 2mg (40%) for **13**. The products were characterized by ¹H-NMR spectroscopy.

Synthesis of the unprotected nitriles 14, 15 and 16

15mg of **11** or 5mg of **12** or 2mg mg of **13** respectively were dissolved in 50 mL of EtOH and 0.5 mL of concentrated sulphuric acid were added. The mixture was heated in a sealed tube for 2 h to 100 °C. Once cooled down to room temperature the reaction mixture was transferred to a separation funnel equipped with 30mL of saturated aqueous NaCl-solution and extracted 4 times with 50mL ethyl acetate EtOAc each. The organic phase was dried over MgSO₄, filtered and concentrated on a rotary evaporator. The pure fractions containing the desired products were evaporated yielding 12mg (80%) of **14**, 3.8mg (75%) of **15** and 1.8 mg (90%) of **16** respectively.

UV-Vis measurements were carried out on a *CARY 5G* spectrophotometer in solution and solid state. The donor acceptor films of (CB4TA), (C2B4TA), (C2,5B4TA) were prepared by thermal evaporation (high vacuum sublimation) at 1x10⁻⁷ atm of pressure at the same time as the photovoltaic devices. The deposit rate of each molecule was 0.5 Å·s⁻¹ and the film thickness was 100 Å for donor acceptor molecules CB4TA, C2B4TA, C2,5B4TA while for the fullerenes the film thickness was 400 Å for the development of multilayer OPVCs [13,14]

The optical energy gap (*E_g*) for all molecules was obtained from the UV-vis spectra using the Tauc approximation [15-18]:

$$(1) \quad E_g = 1242 / \lambda_{\text{onset}}$$

where λ_{onset} is the wavelength threshold obtained from the UV-Vis spectra.

The electrochemical work was performed on a *Voltmaster* model CV 50W potentiostat/galvanostat, using a conventional three-compartment, three-electrode cell, and a polycrystalline platinum disk (0.07 cm² geometric area) as working electrode. The counter electrode was a coiled Pt wire of large area, separated from the electrolytic solution by a sintered glass. The reference electrode was an Ag/AgCl electrode that matches the potential of the saturated calomel electrode (SCE). Then all potentials quoted in the electrochemical section, are referred to the SCE. HOMO and LUMO measurements of each molecule (0.01 mol·L⁻¹) were carried out in anhydrous acetonitrile using tetrabutylammonium hexafluorophosphate (TBAPF₆, 0.1 mol·L⁻¹) as supporting electrolyte. The system was kept under argon atmosphere and all measurements were performed at room temperature [19]

The photovoltaic devices were assembled in a multi heterojunction structure as follows: indium tin oxide (ITO)/ anode buffer layer / donor acceptor molecule / fullerene (C₆₀) / bathocuproine (BCP) / Al [20]. All molecules were used in undoped state. The device was prepared by sublimation at 1×10⁻⁷ atm. The thin film deposition rates and thickness were estimated *in situ* through a quartz monitor. Electrical characterizations were performed with an automated I-V tester, in dark and then emulating solar conditions with a global AM 1.5 simulator. Performances of photovoltaic cells were measured using a calibrated solar simulator (*Oriel* 300W) at 100 mW/cm² light intensity [11,19]

The morphology of the ITO surfaces was characterized by AFM with a Bio AFM JPK in the tapping mode. Measurements were achieved at room temperature, using the same pyramidal Si₃N₄ tip. The surface roughness R_{rms} (root mean square roughness) of each molecular surface was evaluated with the AFM software. [11,19]

The crystalline structure of the films was analyzed by X-ray diffraction (XRD) by a Siemens D 5000 diffractometer using K α radiation from Cu ($\alpha=0.15406$ nm)

3. RESULTS AND DISCUSSION

3.1. Molecular synthesis

The molecules were synthesized via Suzuki-Miyaura cross-coupling reaction [21] according to the procedure described by Bryce et al. [22].

3.2. Molecules characterization

During synthesis optimization thin layer chromatography was used for monitoring the reaction progress. However, after purification of the compounds at those times, the yields found were 70% for CB4TA, 50% for C2B4TA and 30% for C2,5B4TA.

3.3. CB4TA ¹H NMR characterization

¹H NMR (400 MHz, CDCl₃): δ 7.35 (m), 7.25 (d, J = 7.6 Hz, 2H), 7.23 (dd, J = 3.5 and 1.1 Hz, 2H), 7.12 (dd, J = 5.1 and 3.5 Hz, 2H), 6.80 (t, J = 7.62 Hz, 1H), 4.37 (s, 2H).

3.4. C2B4TA ¹H NMR characterization

¹H NMR (400 MHz, CDCl₃): δ 7.32 (m), 7.20 (s, 2H), 7.15 (td, J = 7.9, 7.7 and 1.5 Hz, 2H), 6.81 (td, J = 7.6, 7.5 and 1.2 Hz, 2H), 6.78 (dd, J = 8.0 and 0.9 Hz, 2H), 3.99 (s, 4H).

3.5. C2,5B4TA ¹H NMR characterization

¹H-NMR (400 MHz, CDCl₃): δ 7.20 (m) 7.55 (s, 2H); 6.72 (d, J= 7.5 Hz, 2H); 7.07 (dd, J=7.0 and 1.5Hz, 1H); 7.20 (d, J= 3.5, 1H); 7.22 (d, J= 4.5 Hz, 1H); 7.47 (d, J=7.5Hz, 2H)

3.6. Electrochemical Properties (HOMO LUMO and Eg)

The electrochemical HOMO, LUMO and Eg were determined from solutions of each molecule with concentrations of 0.01 mol·L⁻¹ and TBAPF6 0.1 mol·L⁻¹ supporting electrolyte.

The measurement was done by cyclic voltammetry (CV) (potentials between -2.0 and 1.5 V, 50 mV·s⁻¹ scan rate). Electronic molecule responses were assessed applying a perturbation to the modified electrode (previously to cyclic voltammetry the electrode was coated by the molecules) in anhydrous acetonitrile with TBAPF6 0.1 mol·L⁻¹ free of molecules finding in all cases p-type doping and n-type doping. Figure 2 displays the CV of all molecules.

To estimate the HOMO and LUMO energy, the slope change of the anodic and cathodic current were determined. Figure 2. These values correspond to the onset oxidation potential (E_{ox}), and onset reduction potential (E_{red}), and they correlate linearly with the HOMO and LUMO energy, with a correction factor of 4.4 eV [23-30].

$$E_{\text{HOMO}} = -((E_{\text{ox}})_{\text{on}} + 4.4)\text{eV} \quad (2)$$

$$E_{\text{LUMO}} = -((E_{\text{red}})_{\text{on}} + 4.4)\text{eV} \quad (3)$$

The electrochemical band gap was calculated using the difference between the electrochemical HOMO and LUMO obtained [31]. The optical Eg is in agreement with the electrochemical band gap but with the difference that charge transport in CV is coupled to mass transfer while in optical measurements only electron movement is necessary. Results are shown in table 1 [32-33].

Geometry of the molecules deposited on the platinum electrode is similar to that in solution but, when the Eg is determined by UV-vis in solid state, after desposal by high vacuum sublimation over ITO modified with the buffer layers geometry changes and a red shift is observed Figure 3. Results will be discussed in the UV-vis section. Molecular geometry is very important to obtain desired values for Eg because it plays a decisive role for the conjugation of the π-system on the molecule backbone [19]. Thus changes of the geometry will affect also the conjugation and in consequence the Eg energy.

In molecules with poor conjugation the energy difference of HOMO and LUMO is over 3eV. Also for molecules with in principle good conjugation but with non-coplanar π-systems or high degrees of torsion the conjugation tends to decrease resulting in an increase of the Eg energy. For photovoltaic applications the difference between the energy of frontiers orbitals must be 1-2 eV [34] (molecular geometry and crystallinity will be discussed in XRD analysis).

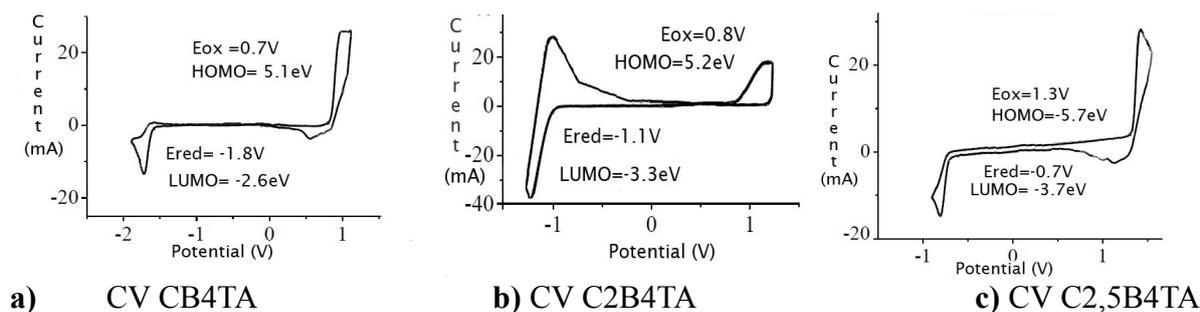


Figure 2. Electrochemical calculation of HOMO LUMO: a) CB4TA b) C2NB4TA c) C2,5NB4TA

Table 1. Electrochemical and optical band gap of molecules

Molecule	HOMO (eV)	LUMO (eV)	E_g (eV)	E_{gopt} (eV) in solution	E_{gopt} (eV) in solid state
CB4TA	-5.1	-2.6	2.5	2.1	2.3
C2B4TA	-5.2	-3.3	1.9	1.8	1.4
C2,5B4TA	-5.7	-3.7	2.0	2.1	1.4

3.7. UV vis spectroscopy

Figure 3 shows UV vis spectra of all molecules in solution (DMSO) and in solid state. It is possible to observe how the electron withdrawing groups produces a red shift and an E_g reduction in both solution and solid state. In solid state is possible to observe a red shift for all materials. This means that conjugation has increased probably due to interactions with buffer layers. In early works we have observed [11,19,34], as well other researchers [35-37] that the CuI buffer layer can interact with molecules with nitrogen in the structure resulting in an ordered first layers of molecules that increases conjugation and decreases the band gap. The affinity of trivalent nitrogen to copper leads to Cu-N interactions and forces the geometry of the molecule. [11,19,34-37].

On the other hand it is possible to observe that nitro groups produce red shift in this molecules molecules.[19,38,39] The reason is probable the decrease of the LUMO energy, which increase the electron affinity and decreases the reduction potential. A more stable LUMO with less energy results in a lower excitation energy when HOMO energy is not changed at the same time. Thus HOMO-LUMO transitions can be observed at lower wavelenghts in UV-vis spectra. As the stabilization effect of the nitro groups is stronger on the LUMO than on the HOMO this explains the observed red-shift.

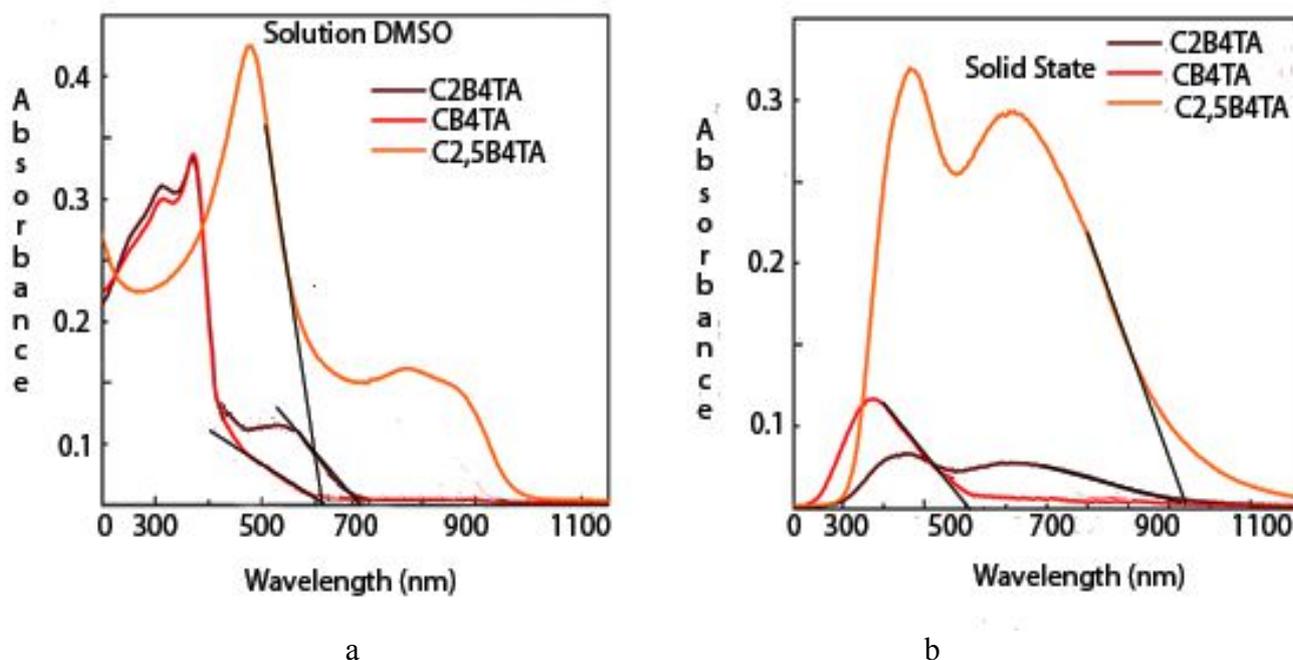


Figure 3. a) UV-vis spectrum in solution b) UV-vis spectrum in solid state

Figure 3b shows that the molecule CB4TA in solid state has a large E_g which can reduce the photovoltaic yield because there are fewer electrons available for excitation to LUMO levels. In this sense the nitro groups help the OPVC to work with low energy photons.

The UV vis spectra show how a large E_g in a molecule with can be reduced by the introduction of nitro groups. This shows that the nitro groups allow to manipulate certain properties in the molecules as well as in OPVC to increase the photovoltaic yield.

3.8. XRD spectroscopy

The XRD shown in figure 4 are certain crystalline domains in the internal layer especially the layers that have contact with the buffer layer. In earlier studies Zamora et al[11,19,34], as well other researchers [35,36] have observed that the buffer layers (CuI) can produce an interaction with polymers containing nitrogen atoms in direct contact ordering the molecules in the first layers as mentioned above. Crystallinity is lost in the rest of the layers, which is visible in the amorphous zones in XRD. This effect has no important influence on the photovoltaic yield because the layers of donor acceptor molecules are slim and the exciton does not have to move over a large distance (100 Å) to arrive at the interface where charge separation occurs. The electron-donor molecule without nitro groups (CB4TA) does not work in the device seen in figure 6. The reason is probably too great quenching when donor and acceptor moieties can transfer charge between them and the excitons “turn off”.

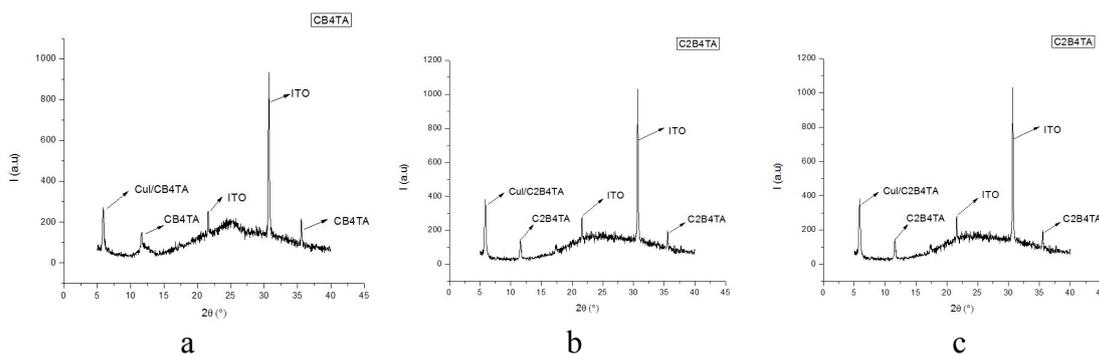


Figure 4. XRD of all molecules deposited over ITO by high vacuum technique (thermal evaporation) a) CB4TA b) C2B4TA c) C2,5B4TA.

3.9. AFM microscopy

Figure 5 shows the interface of donor acceptor molecules deposited on ITO MoO₃ and CuI. The findings are in agreement with XRD results. As explained in the XRD section there exist crystalline zones in the internal layers and AFM microscopy confirms that crystallinity tends to decrease with the distance to the buffer layers. The high roughness of all deposits indicates the absence of crystalline zones in the last layers. This result is very important as high roughness allows more contact surface and improves the exciton dissociation. As shown in Figure 5 the roughness variation of the molecular films influences its optical properties, which will influence the final electrical characteristics of the solar cell. In this sense the increase of the J_{sc} respect to the roughness figure 5 and 6, could be related to a better exciton collection at the p-n junction (interface of electron donor molecules and fullerene) when the molecular surface is smoother producing enhancement of reflectivity of the polymeric film [37].

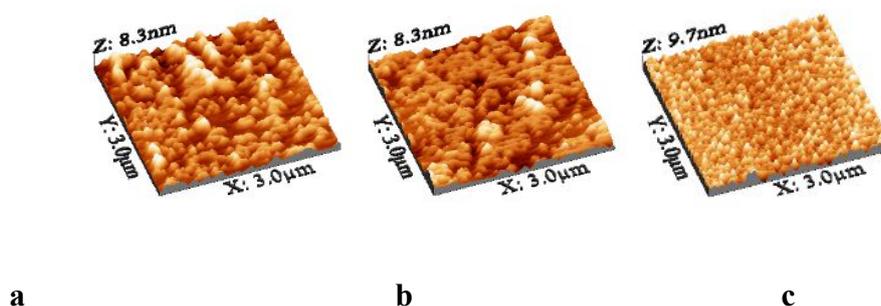


Figure 5. a) AFM of CB4TA rms: 9.5 nm b) AFM of C2B4TA rms: 13.4 nm c) AFM C2,5B4TA rms: 12.5 nm

On the other hand it is possible to see that all peaks have similar shapes. In other words if the deposit of molecules is not uniform the interface could have zones of low roughness which tends to

decrease the contact surface with fullerene. Therefore it is desirable that the last layers have no crystalline zones, allowing high roughness and herewith a large contact area [19].

3.10. Photovoltaic devices

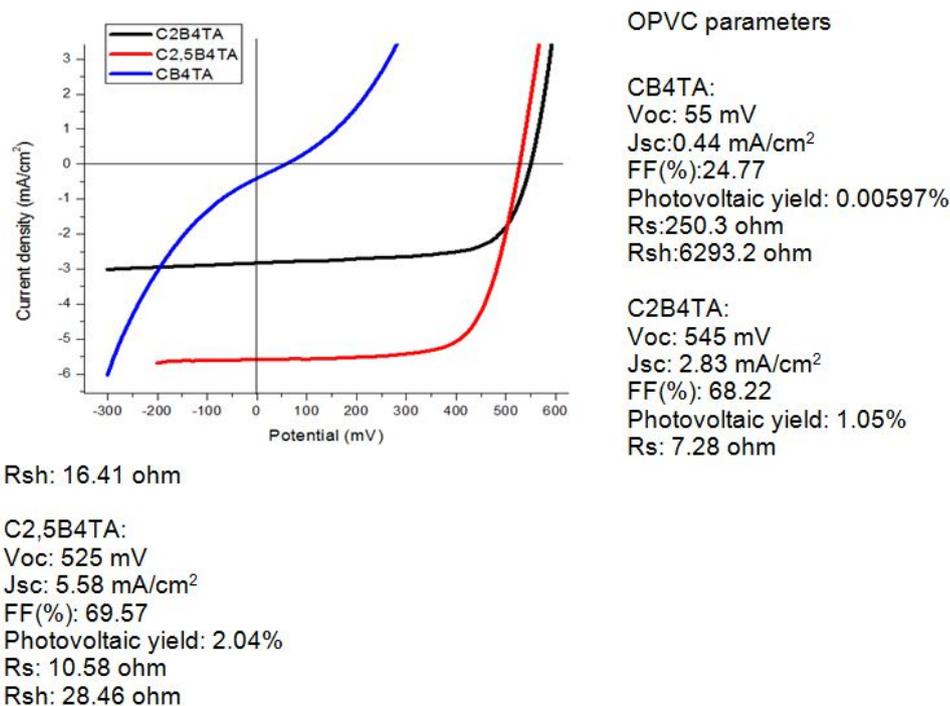


Figure 6. Photoelectrical behavior of OPVCs and cell parameters

Figure 6 shows the behavior of OPVC under solar simulation. The curves for current density (Jsc) and potential (V) confirm that the OPVC using CB4TA has a poor photovoltaic yield and is not useful as solar cell. However in presence of nitro-groups in C2B4TA and C2,5B4TA the photovoltaic yield increases to 2%. The energy level of LUMO decreases which allows the excited electrons (excitons) to pass from HOMO to LUMO more easily and the OPVC can use lower energy photons.

Polarization of the molecule by nitro groups increases molility of the excitons and also intermolecular charge transfer. Transition speed and hole-charge separation are important to prevent exciton quenching [40]. We have shown that use of MoO₃ as buffer layer produces band matching between the HOMO of the first layers of deposits and the HOMO of anode (ITO) [11,19]. In other words MoO₃ approximates the HOMO energy of both, anode and the first molecular layers and this facilitates charge movement to the anode.

The curves J v/s V show low electrical resistance Rs and Rsh. In other words the amorphous zones do not increase the electrical resistance for these molecules. This is probably due the fact that excitons have no large distance to cover to reach the molecule-fullerene interface (layer thickness 100 Å). It is well known that a limiting factor of the yield in organic OPVCs is the electrical resistance of

the organic films. Therefore a decrease of this resistance (R_s and R_{sh}) induces higher short circuit current density and therefore better photovoltaic yield.

4. CONCLUSIONS

The molecules studied in this manuscript have donor and acceptor moieties, separated by a bridge of three thiophene subunits. This structure produces polarization in the molecule due to charge difference between electron donating and electron withdrawing functional groups. The introduction of nitro groups increases the polarization attracting negative charges and facilitating the movement of excitons. Accordingly, molecules bearing nitro groups show a photovoltaic yield close to 2% while the same structure without nitro groups shows a photovoltaic yield not suitable for use as OPVC. This fact shows how nitro groups and possibly other electron withdrawing groups can facilitate the movement of excitons decreasing the probability of quenching in OPVC. Thus chemical modifications with electron acceptor groups of molecules with poor photovoltaic behavior tends to decrease the LUMO energy level. Thus the molecules can act as electron donors in OPVCs.

On the other hand thin of electron donor films can interact with CuI as shown XRD data, forming areas of crystallinity, which facilitate the movement of excitons and charges. Using AFM one can observe the high roughness at the interface between the thin layer of electron donor molecules and the fullerene as electron acceptor.

The introduction of nitro groups and other electron acceptors should produce similar effects on other families of molecules. Thus electron withdrawing groups should be taken into account when looking for strategies to improve photovoltaic yields for different substrates in their application in OPVCs.

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