

Deposition of carbon nanotubes onto aramid fibers using as-received and chemically modified fibers



O. Rodríguez-Uicab^a, F. Avilés^{a,*}, P.I. Gonzalez-Chi^a, G. Canché-Escamilla^a, S. Duarte-Aranda^a, M. Yazdani-Pedram^b, P. Toro^c, F. Gamboa^d, M.A. Mazo^e, A. Nistal^e, J. Rubio^e

^a Centro de Investigación Científica de Yucatán A.C., Unidad de Materiales, Calle 43 No.130, Col. Chuburna de Hidalgo, C.P. 97200 Mérida, Yucatán, Mexico

^b Facultad de Ciencias Químicas y Farmacéuticas, Universidad de Chile, S. Livingstone 1007, Independencia, Santiago, Chile

^c Facultad de Ciencias Físicas y Matemáticas, Universidad de Chile, Av. Beauchef 850, Santiago, Chile

^d Centro de Investigación y de Estudios Avanzados del IPN, Unidad Mérida, Depto. de Física Aplicada, Km. 6 Antigua Carretera a Progreso, 97310 Mérida, Yucatán, Mexico

^e Instituto de Cerámica y Vidrio (ICV-CSIC), Kelsen 5, 28049 Madrid, Spain

ARTICLE INFO

Article history:

Received 9 April 2016

Received in revised form 5 May 2016

Accepted 6 May 2016

Available online 10 May 2016

Keywords:

Aramid fiber

Carbon nanotubes

Surface treatment

Mechanical properties

Electrical properties

Multifunctional

ABSTRACT

Multiwall carbon nanotubes (MWCNTs) oxidized by an acid treatment were deposited on the surface of as-received commercial aramid fibers containing a surface coating (“sizing”), and fibers modified by either a chlorosulfonic treatment or a mixture of nitric and sulfuric acids. The surface of the aramid fiber activated by the chemical treatments presents increasing density of CO, COOH and OH functional groups. However, these chemical treatments reduced the tensile mechanical properties of the fibers, especially when the nitric and sulfuric acid mixture was used. Characterization of the MWCNTs deposited on the fiber surface was conducted by scanning electron microscopy, Raman spectroscopy mapping and X-ray photoelectron spectroscopy. These characterizations showed higher areal concentration and more homogeneous distribution of MWCNTs over the aramid fibers for as-received fibers and for those modified with chlorosulfonic acid, suggesting the existence of interaction between the oxidized MWCNTs and the fiber coating. The electrical resistance of the MWCNT-modified aramid yarns comprising ~1000 individual fibers was in the order of $M\Omega/cm$, which renders multifunctional properties.

© 2016 Elsevier B.V. All rights reserved.

1. Introduction

Aramid fibers are well known for their use in ballistic scenarios and in advanced composites for impact-related applications, such as bulletproof vests, helmets, and high performance composites for transportation and the automobile industry [1,2]. These fibers do not show a melting point and their thermal decomposition occurs near 400 °C [2–4]. The aromatic rings present in the backbone of the fiber molecular structure promote high thermal stability [5]. Aramid fibers comprise a highly ordered arrangement of polymer chains yielding a rather smooth surface with reduced reactivity [3–7], high crystallinity (~76–95%) and high mechanical strength [8,9]. However, in the composites materials field, surface modifications of the aramid fibers are often necessary to activate their surface. There are several surface treatments that have been

attempted to promote the generation of functional groups on the surface of aramid fibers, including plasma treatments [10], acid chemical treatments [11] and fluorinations [12]. Wu et al. [13], for example, investigated the effect of fiber surface treatments using plasmas of ammonia, oxygen and water vapor, as well as a solution of chlorosulfonic acid in dichloromethane. According to their results, the chlorosulfonic acid treatment produced more changes to the fiber surface morphology than the plasma treatments. Maity et al. [14] reported a modification of aramid fibers using fluorination; the C-H bonds of the aromatic rings were substituted by C-F bonds, yielding higher thermal stability. In multiscale hierarchical composites comprising a macro-scale matrix, a fiber with diameter at the micro-scale, and a nanostructure, these kind of surface modifications may improve the interactions not only between the fiber and matrix, but also with the nanostructures deposited onto the fiber surface, such as carbon nanotubes. The current interest in multiscale hierarchical composites focuses on their multifunctionality, and in the case of multiwall carbon nanotubes (MWCNTs) their high electrical conductivity is frequently exploited. MWCNTs

* Corresponding author.

E-mail address: faviles@cicy.mx (F. Avilés).

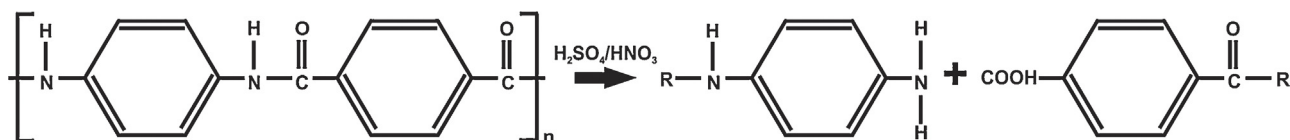


Fig. 1. Proposed hydrolysis reaction for aramid fibers treated with the mixture of nitric and sulfuric acids.

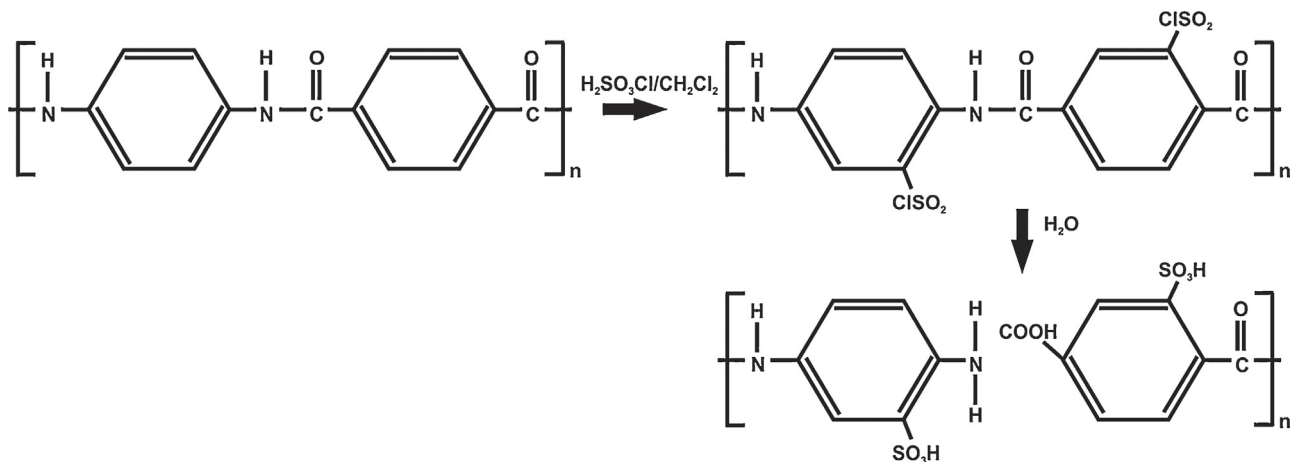


Fig. 2. Proposed reaction for aramid fibers treated with chlorosulfonic acid.

have been deposited onto glass and carbon fibers by electrophoretic methods [15,16] and by simpler methods such as dipping or immersion in nanotube dispersions [17–19]. It has been shown that the deposition of MWCNTs onto engineering fibers can increase the fiber/matrix interfacial shear strength [20] and delay the onset of microcrack propagation [21]. Furthermore, the generation of a percolative electrical networks on the fiber surface renders composite materials with multifunctional features such as self-sensing of strain and damage [20]. However, research concerning deposition of MWCNTs onto aramid fibers is scarce. O'Connor et al. [22] reported the deposition of MWCNTs onto aramid fibers using *N*-methylpyrrolidone (NMP). The aramid fibers and the MWCNTs were immersed in NMP to swell the fibers, promoting physical interactions between the aramid fibers and the MWCNTs. A similar work was reported by Chen et al. [3], where the aramid fibers were modified by a mixture of hexamethylene diisocyanate and 1,4-diazabicyclooctane. The MWCNTs were dispersed in NMP and the aramid fibers were immersed in the MWCNT/NMP solution, promoting fiber swelling and chemical grafting. Given this background and motivation, MWCNTs were herein deposited onto the surface of Twaron fibers using an immersion process assisted by ultrasound; the role of the surface coating of the Twaron fibers was studied examining the effect of two fiber chemical treatments.

2. Materials and methods

2.1. Materials

The aramid fibers used were Twaron 2200 from Teijin Aramid Inc. (Georgia, USA). According to the manufacturer, the fibers have a tensile elastic modulus between 130 and 180 GPa, a density of 1.45 g/cm³ and individual fiber diameter of ~12 μm. The fiber yarn contains ~1000 individual filaments. After their commercial synthesis, the fibers are coated with chemical agents generically called “fiber surface coating (FSC)” and frequently referred to in the jargon as “sizing” or “finish”. This coating reduces fiber damage when handled and improves the processability of the fibers [23]. The FSC formulation is complex and varies according to the final applica-

tion and manufacturer; it may contain, among others, additives, lubricants, antistatic agents, emulsifiers and antioxidants [23].

Commercial MWCNTs were acquired from Cheap Tubes Inc. (Vermont, USA), with a typical length of 1–6 μm, an inner average diameter of 5–15 nm and an outer average diameter of 30–50 nm [24]. Chemical oxidation of the MWCNTs was conducted with a mixture of H₂SO₄ (97.9% v/v) and HNO₃ (65.4% v/v) from J.T. Baker (Pennsylvania, USA). Chlorosulfonic acid at 97% v/v from Merck & Co (New Jersey, USA) and dichloromethane 99.8% v/v from Winkler LTDA (Santiago, Chile). To remove the FSC, ethanol and methanol 98% v/v were acquired from Merck, acetone (98% v/v) from Winkler LTDA and dichloromethane 99.8% v/v also from Winkler LTDA. The deposition of MWCNTs onto the fiber surface was made with Chloroform at 99.9% v/v from J.T. Baker.

2.2. Chemical oxidation of carbon nanotubes

All MWCNTs used in this work were chemically oxidized. A 3.0 M mixture of nitric and sulfuric acids was used for chemical oxidation of MWCNTs, using a previously reported method [25]. Briefly, the acid mixture and MWCNTs were first mechanically stirred in a hot plate for 15 min at ~60 °C. Then, the acid mixture and MWCNTs were ultrasonically dispersed using an ultrasonic bath of 70 W and 42 kHz for 2 h. The slurry was then filtered, thoroughly washed with distilled water and dried at 100 °C for 24 h.

2.3. Aramid fiber treatments

The aramid fiber yarns used are classified into three groups, viz. as-received (without treatment) and treated by one of the two chemical modifications conducted herein. Prior to these chemical treatments, the FSC was removed by Soxhlet extraction and sequential immersions in solvents, as described in Section 2.3.1. Then, the aramid fibers were subjected to a treatment with chlorosulfonic acid or to a (more aggressive) treatment using a mixture of nitric and sulfuric acids. Finally, previously oxidized MWCNTs were deposited on each of the three types of aramid fibers using a chloroform immersion method assisted by ultrasound. The following

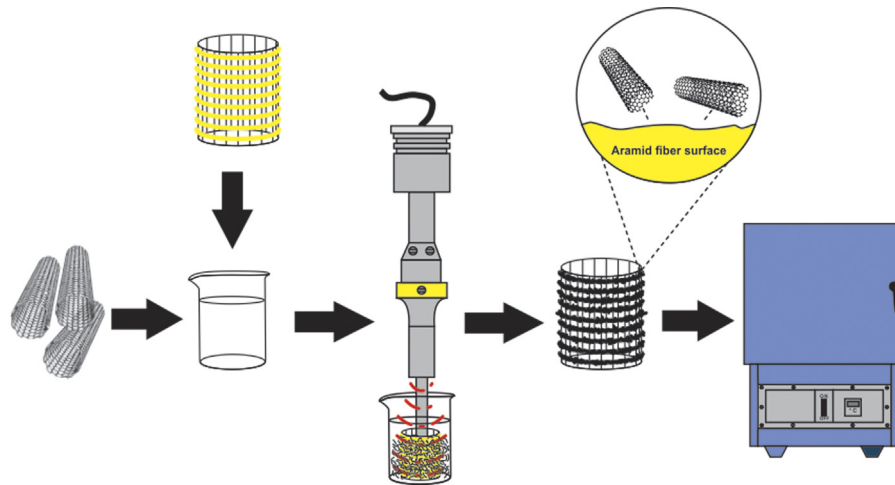


Fig. 3. Schematic of the method used for the deposition of MWCNTs onto aramid fibers.

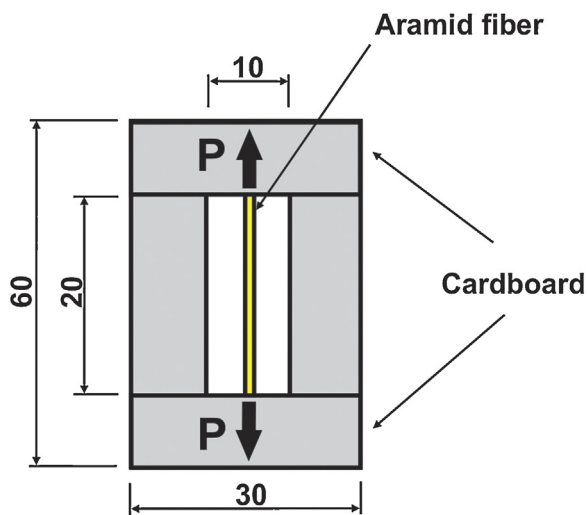


Fig. 4. Setup used for tensile testing of individual fibers. Dimensions in mm. *P* shows the loading direction.

sections describe the removal of the FSC, the chemical treatments and the MWCNT deposition processes.

2.3.1. Removal of fiber surface coating

The FSC was removed by a Soxhlet extraction and sequential immersions in chloroform, ethanol, acetone, and methanol, as recommended in [4]. The aramid fibers were dried in a convection oven at 70 °C after each sequential step. First, the aramid fibers were placed in a Soxhlet extractor recirculating chloroform for 6 h. The process was repeated in the Soxhlet extractor for 6 h using ethanol, and then with acetone and methanol. After removing the FSC, one of the two chemical treatments described below (Sections 2.3.2 and 2.3.3) were applied to activate the fiber surface.

2.3.2. Fiber treatment by a mixture of nitric and sulfuric acids

The modification of the aramid fiber surface (with FSC previously removed, see Section 2.3.1) was performed by immersing the yarn in a 3.0 M mixture of HNO_3 and H_2SO_4 for 1 h. Then, the aramid fibers were washed with 1.5 L of distilled water. This treatment is expected to affect the amorphous phase of the aramid fibers (<24%) through amide bond opening situated at the backbone of the fiber structure generating amide and carboxyl groups in some fiber sections, see Fig. 1 [26,27]. These functional groups can interact

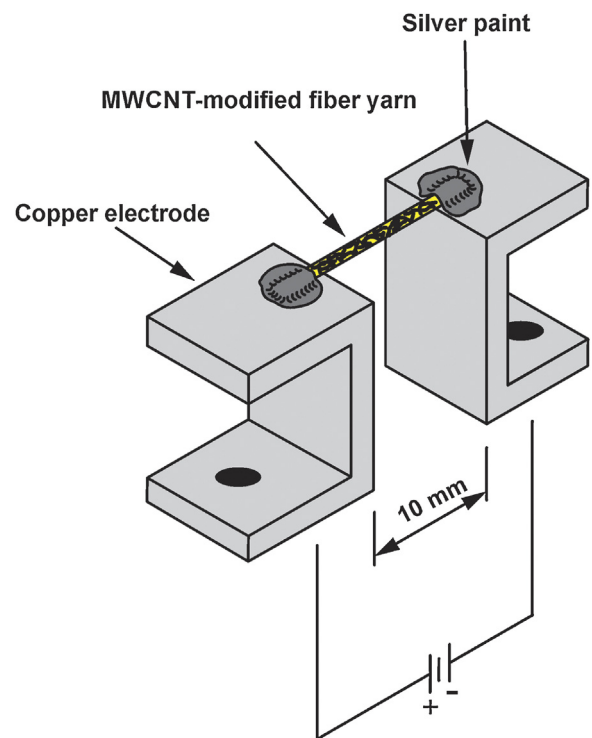


Fig. 5. Setup used to measure the electrical resistance of MWCNT-modified aramid fiber yarns.

through hydrogen bonds with the carboxyl and hydroxyl groups of the previously oxidized MWCNTs [28].

2.3.3. Fiber treatment by chlorosulfonic acid

The aramid fibers whose FSC was previously removed were treated with a solution of chlorosulfonic acid in dichloromethane at 0.2% w/w, stirring the solution at room temperature for 2 min. Subsequently, the aramid fibers were immersed in distilled water stirring for 2 more min and finally dried for 1 h at 70 °C. This treatment is expected to produce sulfonyl chloride (SO_2Cl) groups in the active sites of the aromatic rings, see Fig. 2. The subsequent immersion in distilled water is expected to convert the sulfonyl chloride groups into sulfonic groups (SO_3H), as discussed in [4]. This treatment is also expected to modify the amorphous phase of the aramid fiber surface [26,27]. The central amide group of the

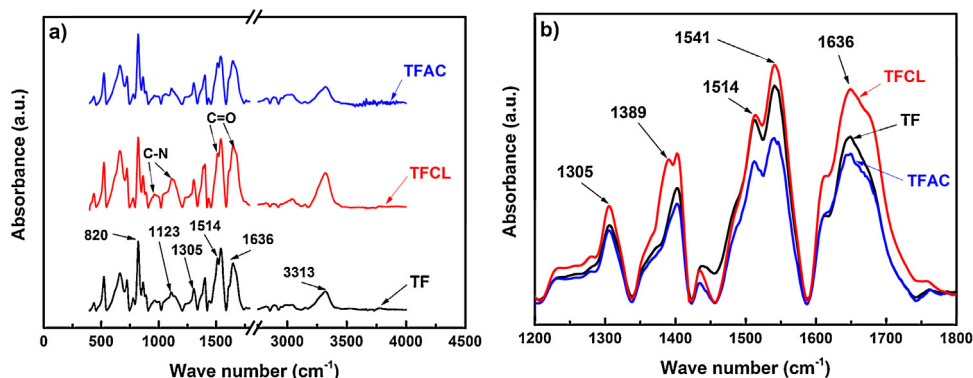


Fig. 6. FT-IR analysis of aramid fibers without and with chemical treatment. a) Spectra of TF, TFCL and TFAC, b) 1200–1800 cm^{-1} region normalized with the 820 cm^{-1} band.

Table 1
Nomenclature used for the aramid fibers without and with MWCNTs.

| Label | Aramid fiber condition |
|----------|---|
| TF | Pristine fibers (as-received). |
| TFAC | Fibers were immersed in a mixture of sulfuric and nitric acids for 3 h. |
| TFCL | Fibers were immersed in a solution of chlorosulfonic acid in dichloromethane for a few minutes. |
| TF-CNT | TF with deposited MWCNTs. |
| TFAC-CNT | TFAC with deposited MWCNTs. |
| TFCL-CNT | TFCL with deposited MWCNTs. |

fiber can separate due to sulfonation yielding carboxylic functional groups (see Fig. 2), which are expected to interact with the OH, COOH and CO functional groups of the oxidized MWCNTs through hydrogen bonding [28].

2.4. Deposition of carbon nanotubes onto aramid fibers

Fig. 3 shows the process used to deposit MWCNTs onto aramid fibers. 4 mg of previously oxidized MWCNTs were deposited onto ~550 mg of aramid fibers rolled on a cylindrical frame and immersed into 100 mL of chloroform. The MWCNT deposition onto aramid fibers was conducted by using an ultrasonic horn for 1 h at 165 W and 20 kHz. Finally, the aramid fibers were dried in a convection oven for 2 h at 100 °C.

The nomenclature of the as-received, chemically modified and fibers with deposited MWCNTs used in the present study are described in Table 1.

2.5. Fiber characterization

2.5.1. Infrared spectroscopy

Fourier transform infrared spectroscopy (FT-IR) was conducted using a Nicolet 8700, Thermo Fisher Scientific spectrometer. The FT-IR spectra of the samples were collected in the 650–4000 cm^{-1} interval with a resolution of 4 cm^{-1} . The samples were analyzed in the absorbance mode using KBr pellets. This analysis was repeated 5 times finding reproducibility.

2.5.2. Raman spectroscopy

Raman spectroscopy of individual aramid fibers fixed in a cardboard frame was obtained by a micro Raman spectrometer equipped with an Andor DV401 CCD camera. A laser of 35 mW with a wavelength of 632.8 nm (1.96 eV) was used. Raman spectra were collected in the frequency range of 20–200 cm^{-1} . The analysis of individual Twaron fibers was conducted by mounting the fiber in a cardboard frame using adhesive tape. The total exposition time was 25 min.

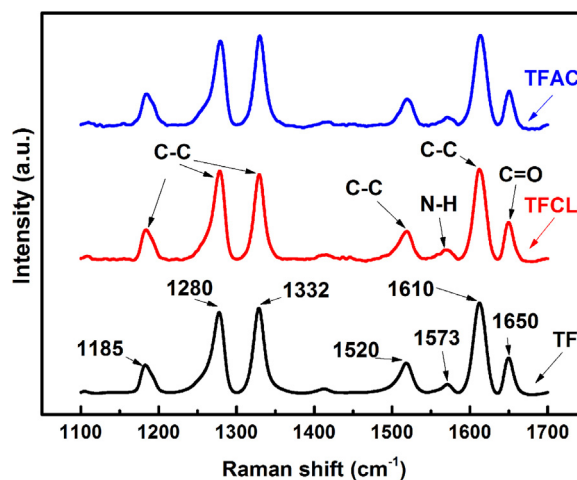


Fig. 7. Raman spectra of as-received and chemically modified aramid fibers.

The analysis of the MWCNT distribution on the fiber was performed using a WITec ALPHA 300R confocal Raman spectrophotometer. Raman spectra of 30×45 pixels were obtained with an exposition time of 2 s per pixel covering an area of $5 \times 10 \mu\text{m}$. For this mapping, each pixel was generated by the intensity of the 2D Raman band of the MWCNTs at $\sim 2700 \text{cm}^{-1}$ [31]. The images were processed using a WITec Project 2.08 software.

2.5.3. X-ray photoelectron spectroscopy

X-ray photoelectron spectroscopy (XPS) was carried out over aramid fiber tows using a sampling area of $400 \mu\text{m}$ by $400 \mu\text{m}$. A Thermo Scientific K-Alpha™ X-ray photoelectron spectrometer with monochromatic Al $K\alpha$ X-ray source operated with an energy of 1486.6 eV. The XPS survey scans were obtained by setting the analyzer to a 1 eV energy pass, while an energy pass of 0.1 eV was employed for high resolution windows. The high resolution C1s orbital curves were corrected by using a baseline obtained from a Shirley background [29,30], and fitted using Voigt functions [29] to obtain additional information.

2.5.4. Scanning electron microscopy

Morphological analysis of the fiber surface and analysis of the presence of MWCNTs on the fibers were performed by scanning electron microscopy. SEM analysis was performed by using a JEOL JSM-630-LV microscope at 20 kV.

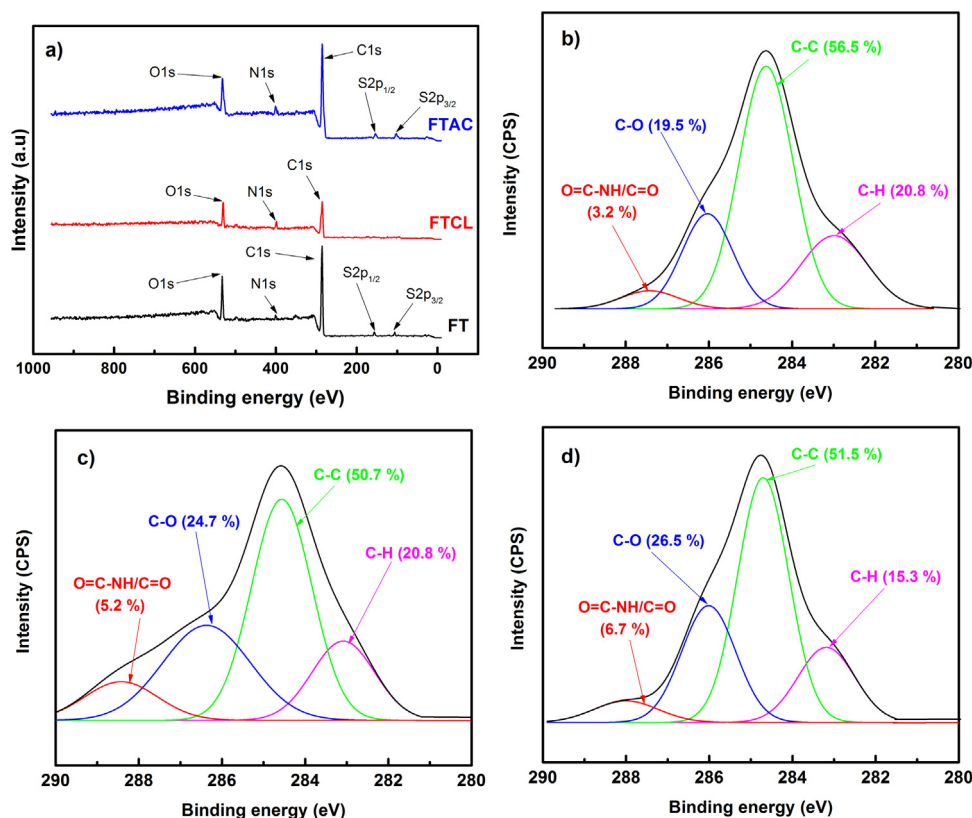


Fig. 8. XPS analysis of as-received and chemically modified aramid fibers. a) Survey, b) C1s orbital of TF, c) C1s orbital of TFCL, d) C1s orbital of TFAC.

2.6. Tensile testing

Tensile testing of individual Twaron fibers was conducted by mounting the fiber in a cardboard frame using adhesive tape and a commercial cyanoacrylate-based glue (Fig. 4). Tests were conducted in an AGS-X Shimadzu universal testing machine using a double sensitivity load cell of 100N at cross-head speed of 1 mm/min. Since strain was measured with the machine cross-head displacement, a compliance correction was carried out for the elastic modulus (E) and maximum strain (ε_{\max}), using specimens with effective gauge lengths of 10, 20 and 40 mm, following the procedure described by Adams et al. [32].

2.7. Electrical resistance measurements

Direct current electrical resistance of MWCNT-modified aramid fiber yarns (comprising ~ 1000 individual fibers) was measured using a two point method. 15 mm long yarns were fixed in a cooper frame with an effective span of 10 mm, as shown in Fig. 5. The yarn ends were bonded to the cooper electrodes using commercial conductive silver paint (Ted Pella Inc. type 16062, CA, USA) and the electrical resistance was measured using a Keithley 7517B electrometer.

3. Results

3.1. Characterization of modified aramid fiber

3.1.1. Infrared spectroscopy

Fig. 6 shows the FT-IR analysis of the as-received Twaron fibers (TF) and those with a chemical modification (TFCL and TFAC),

as described in Table 1. Fig. 6a shows the absorption bands of TF, TFAC and TFCL. The bands at 1123 cm^{-1} and 1305 cm^{-1} are attributed to stretching vibrations of C-N bonds [33]. The bands at 1514 cm^{-1} and 1636 cm^{-1} are attributed to stretching vibrations of N-H and C=O groups, respectively [33,34]. The band at 1541 cm^{-1} is related to the combined motion of N-H bending and C-N [23]. The band at 3313 cm^{-1} is related to stretching vibrations of N-H of the amide groups [34]. Additionally, the band located at 820 cm^{-1} is assigned to stretching vibrations of the C-H bonds of the aromatic ring present in the backbone structure of the aramid fiber. This band does not present significant variations in intensity when the fiber is subjected to chemical treatments [23], and therefore was selected for normalization purposes. A detailed analysis in the interval $1200\text{--}1800\text{ cm}^{-1}$ is included in Fig. 6b, where the intensities have been normalized with the intensity of the C-H band located at 820 cm^{-1} .

Fig. 6b shows a slight increase in the intensity of the band at 1305 cm^{-1} (C-N) for TFCL with respect to TF, which can be attributed to the sulfonation of the aromatic rings [35]. Further indication of sulfonation is observed from a new band of small intensity at 1389 cm^{-1} for FTCL. The band at 1636 cm^{-1} also shows an increase in intensity for TFCL compared to TF, indicating fiber oxidation. The slight decrease in the C=O band at 1636 cm^{-1} for TFAC with respect to TF is possibly caused by a competition between fiber oxidation and rupture of the bonds between the C=O and the N-H groups of the main aramid chain for such an aggressive fiber treatment [23]. Further indication of rupture of the amide bonds for TFAC is observed by the intensity decrease of the bands at 1514 cm^{-1} and 1541 cm^{-1} , corresponding to N-H [35].

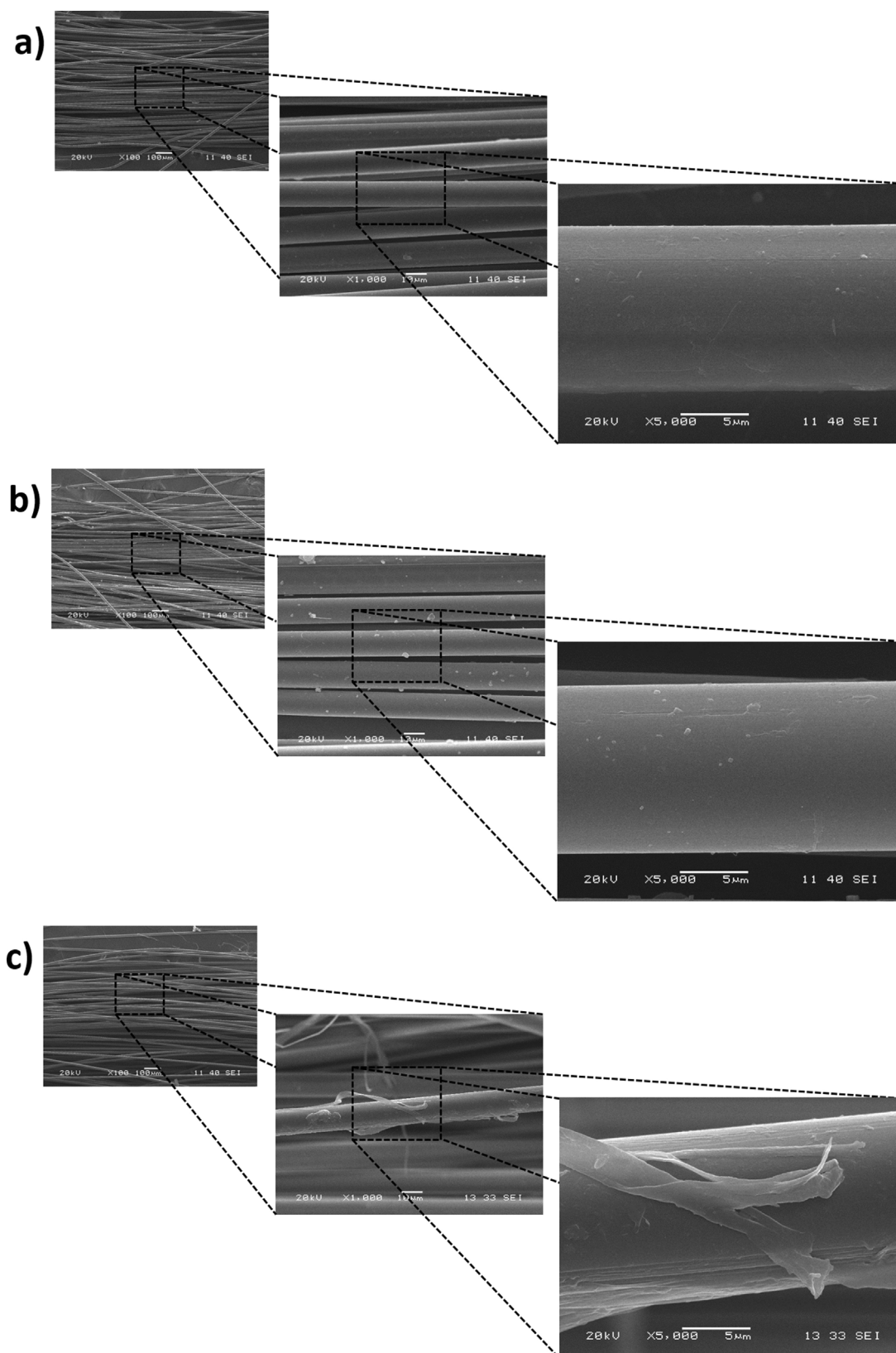


Fig. 9. SEM micrographs of the as-received and treated aramid fibers. a) TF, b) TFCL, c) TFAC.

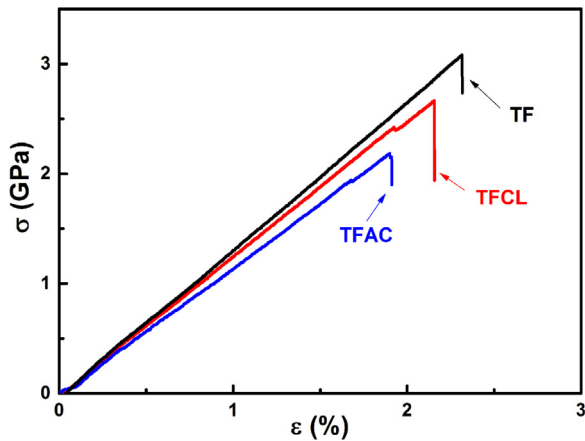


Fig. 10. Representative stress-strain curves of individual aramid fibers TF, TFCL and TFAC.

3.1.2. Raman spectroscopy

Fig. 7 shows the results of Raman spectroscopy for TF, TFCL and TFAC. Fig. 7a shows bands at 1185, 1280, 1332, 1520 and 1610 cm^{-1} related to stretching vibrations of the C-C bonds in the aromatic rings of the structure of the aramid fiber [36]. The band at 1571 cm^{-1} is attributed to stretching vibrations of the N-H bond. The vibrations at 1652 cm^{-1} are attributed to stretching modes of the C=O groups present in the main fiber structure [36]. The spectra of TFCL and TFAC are similar to that of TF, suggesting that the chemical treatments carried out do not significantly modify the main structure of the aramid fiber. Since the Raman spectroscopy signal of the TF fiber is very intense (given its high crystallinity) and only a small amorphous phase at the fiber surface is expected to be modified by the chemical treatments, a surface change in the TF is difficult to determine by this technique.

3.1.3. X-ray photoelectron spectroscopy

Fig. 8 shows the XPS spectra of TF, TFCL and TFAC. These spectra show three high intensity bands corresponding to C1s, N1s and O1s orbitals at ~ 285 , 400 and 532 eV, respectively [34,37]. For TF, small bands assigned to sulfur at 105 and 156 eV are observed, which may stem from traces of the sulphuric acid employed in its synthesis process [5,37]. On the other hand, the bands at 105 and 156 eV in the TFAC fiber are generated by the acid treatment of the fiber with sulfuric acid, as TFCL fibers do not exhibit such bands. For TFCL and TFAC a slight increase in intensity of the N1s orbital with respect to that of TF is observed. This increase is attributed to the removal of the FSC, which causes that a larger amount of nitrogen bonds become exposed on the fiber surface [6,38].

For a more detailed analysis, Figs. 8b–d show deconvolutions of the C1s orbital (~ 285 eV) for all treatments, showing four bands; the first one from left to right located ~ 287.8 eV corresponds to the contribution of O=C-N-H and C=O bonds (3.2%), coming from the amide and carbonyl functional groups present in the molecular structure of the aramid fiber [6]. Bands at ~ 286.0 and 284.6 eV are associated to C-O (19.5%) and C-C (56.5%) bonds, respectively, coming from the FSC and aromatic rings of the fiber [6,38]. Figs. 8c and d show a deconvolution of the C1s orbital for TFCL and TFAC fibers, respectively. Because of the surface treatments, the proportions of the functional groups on the fiber surface significantly change with respect to the untreated fiber. For TFCL and TFAC fibers the contribution of O=C-N-H/C=O groups is larger, since the fiber is uncovered and depleted from the FSC, see e.g. [6,38]. For TF fibers, the contribution of O=C-N-H bonds is 3.2%, while for TFCL and TFAC is 5.2% and 6.7%, respectively; this suggests that the FSC has been removed and that the treatment based on the acid mixture generates more oxygen-containing functional groups, being more aggressive. Evidence of fiber chemical oxidation is observed from the increase of oxygen-containing functional groups such as C-O. These functional groups contribute to the C1s orbital in 19.5% for TF, and their contribution greatly increase to 24.7% for TFCL and 26.5% for TFAC. This

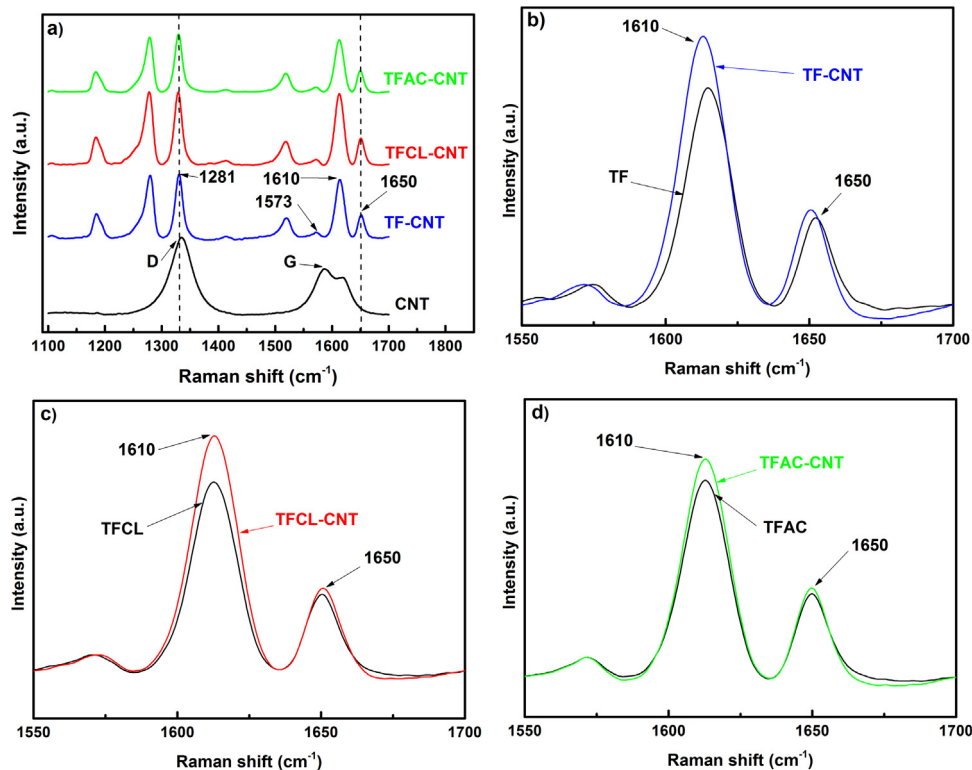


Fig. 11. Raman spectra of the aramid fibers. a) MWCNTs (CNT), TF-CNT, TFCL-CNT and TFAC-CNT, b) TF and TF-CNT, c) TFCL and TFCL-CNT, d) TFAC and TFAC-CNT.

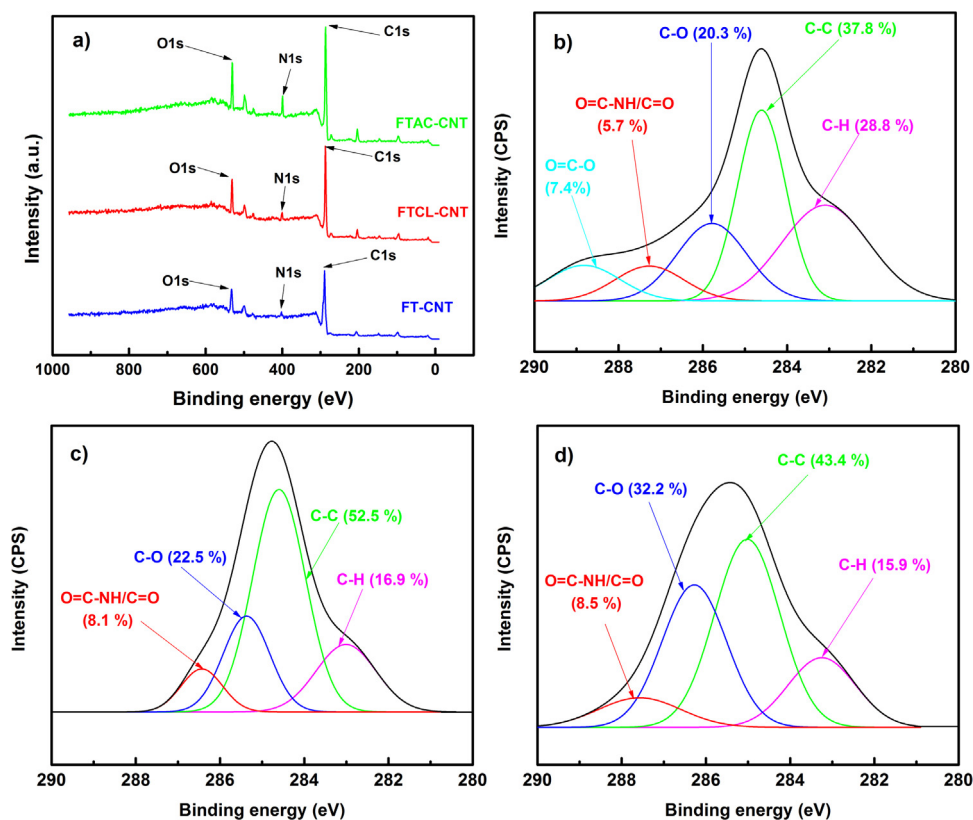


Fig. 12. XPS spectra of MWCNT-modified aramid fibers. a) Surveys, b) C1s orbital of TF-CNT, c) C1s orbital of TFCL-CNT, d) C1s orbital of TFAC-CNT.

Table 2

Tensile properties of as-received and chemically modified aramid fibers.

| Fiber | E (GPa) | σ_{max} (GPa) | ϵ_{max} (%) |
|-------|----------------|----------------------|----------------------|
| TF | 167 ± 25.3 | 2.70 ± 0.33 | 2.47 ± 0.50 |
| TFCL | 134 ± 10.8 | 2.56 ± 0.31 | 2.01 ± 0.22 |
| TFAC | 124 ± 5.0 | 1.97 ± 0.40 | 1.76 ± 0.29 |

indicates that the TFAC fiber has higher degree of oxidation than TF and TFCL.

3.1.4. Scanning electron microscopy

Fig. 9 shows SEM micrographs of aramid fiber yarns taken at 100x, 1000x and 5000x magnifications (from left to right). The as-received aramid fiber (Fig. 9a) shows a relative smooth surface with shallow longitudinal markings produced by their synthesis process. TFCL fibers (Fig. 9b) show similar characteristics to TF but they show new surface markings and the presence of some irregular material (particles) on the fiber surface. This material may correspond to traces of the FSC removed by Soxhlet extraction and by the subsequent chemical treatment carried out. The acid treatment (TFAC) was more aggressive with the fiber surface, which is evident in Fig. 9c by the peeling off of some surface layers.

3.1.5. Mechanical properties

Tensile stress (σ) vs. strain (ϵ) curves of the as-received (TF) and chemically modified (TFAC and TFCL) fibers with gauge lengths of 20 mm are presented in Fig. 10. Table 2 summarizes the mechanical properties (average and standard deviation) obtained from such curves, listing the elastic modulus (E), strength (σ_{max}) and the maximum strain (ϵ_{max}).

E , σ_{max} and ϵ_{max} measured for the as-received fibers are similar to that reported in the literature [5]. According to the results of Fig. 10 and Table 2, TF and TFCL show statistically similar mechan-

ical properties, although the ones treated with chlorosulfonic acid show a trend to reduce their mechanical properties. On the other hand, fibers treated with the acid mixture (TFAC) show a more considerable reduction in E , σ_{max} and ϵ_{max} with respect to TF. These results are consistent with the ones obtained from FT-IR, XPS and SEM, all suggesting that the treatment with nitric and sulphuric acids is more oxidative and also damages more the aramid fibers.

3.2. Characterization of MWCNT-modified aramid fibers

3.2.1. Raman spectroscopy

Fig. 11 shows the Raman spectra of MWCNT-modified aramid fibers. Fig. 11a shows the spectrum of the oxidized MWCNTs (bottom, labeled as "CNT") and of the TF, TFCL and TFAC fibers containing MWCNTs on their surface. The typical D and G bands of the MWCNTs are observed at 1335 and 1587 cm^{-1} , respectively [31]. The spectra of the MWCNT-modified fibers TF-CNT, TFCL-CNT and TFAC-CNT show two bands at 1281 and 1610 cm^{-1} corresponding to the C-C vibration of the aromatic rings of the molecular structure of the aramid fiber (see Fig. 7). These bands belonging to the fibers overlap with the D and G bands of the MWCNTs, and consequently, their intensity and width increase when MWCNTs are deposited on the fiber surface, as shown in Figs. 11b–d for the band centered at 1610 cm^{-1} .

For a more detailed analysis of the band at 1610 cm^{-1} , the 1550 – 1700 cm^{-1} spectral range was further investigated (see Figs. 11b–d), normalizing the band intensities with the intensity of the band at 1571 cm^{-1} , which is attributed to N–H vibrations. As observed from these figures, the presence of MWCNTs on the TF-CNT, TFCL-CNT and TFAC-CNT fibers increases the intensity and width of the band at 1610 cm^{-1} , with respect to the fibers without MWCNTs. To quantify this observation the area under the band at 1610 cm^{-1} (A_{1610}) was divided by the corresponding area under

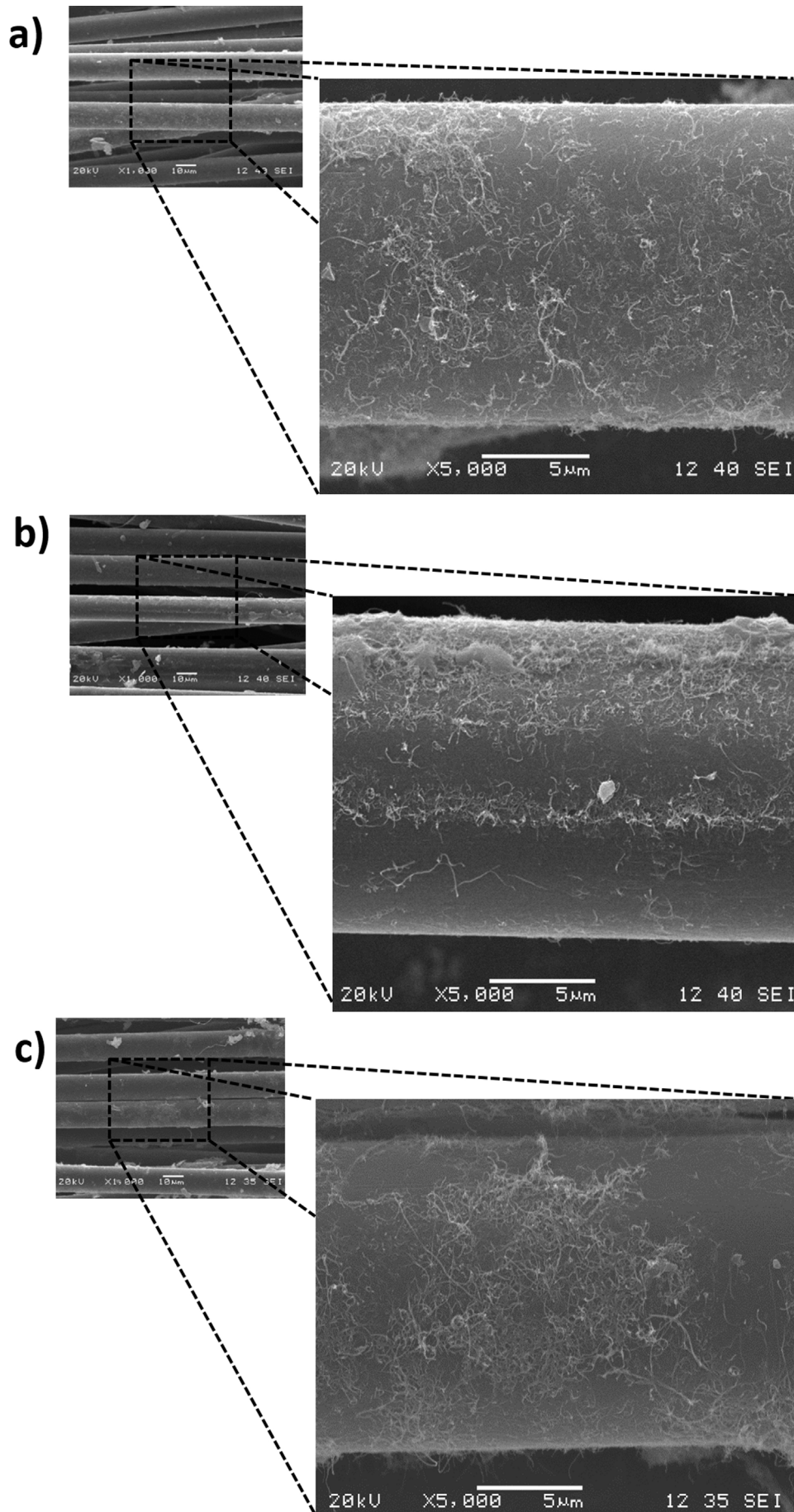


Fig. 13. SEM micrographs of MWCNT-modified aramid fibers. a) TF-CNT, b) TFCL-CNT, c) TFAC-CNT.

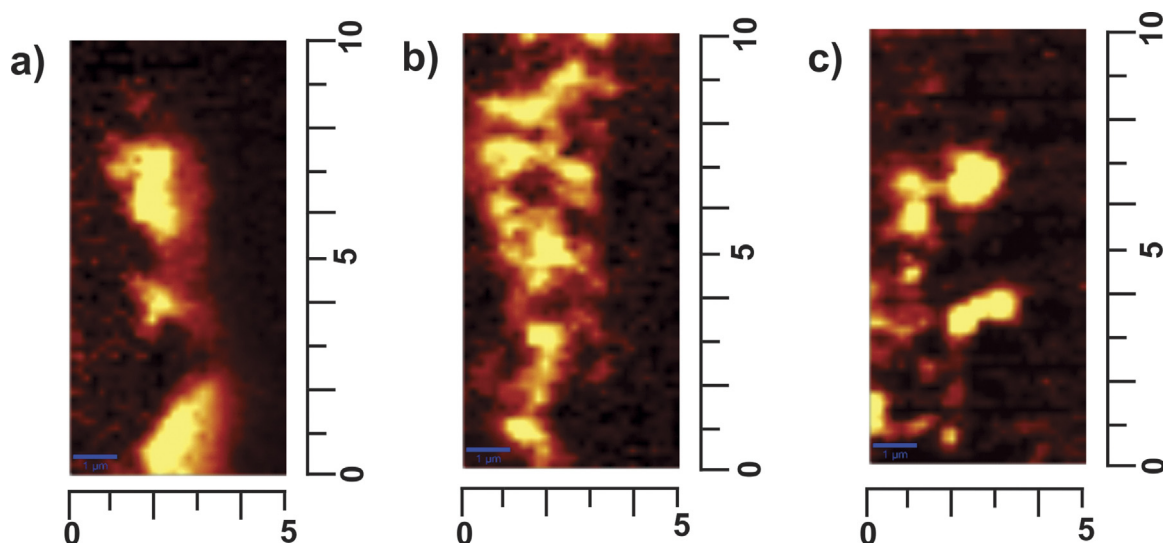


Fig. 14. 2D Raman band intensity maps of the MWCNTs on the aramid fiber surface. a) TF-CNT, b) TFCL-CNT, c) TFAC-CNT. Dimensions in μm .

Table 3

Area under the curve of the band at 1610 cm^{-1} normalized by the area of the band at 1571 cm^{-1} .

| Fiber | A_{1610}/A_{1571} | Area increment with respect to the corresponding fiber without MWCNTs (%) |
|----------|---------------------|---|
| TF | 211 ± 4.6 | – |
| TF-CNT | 254 ± 9.8 | 20.3 |
| TFCL | 238 ± 10.3 | – |
| TFCL-CNT | 284 ± 11.5 | 18.7 |
| TFAC | 190 ± 2.6 | – |
| TFAC-CNT | 208 ± 8.8 | 9.6 |

the band at 1571 cm^{-1} (A_{1571}), taken as reference. Table 3 shows the results of this metric, reporting average and standard deviations corresponding to 5 replicates. The last column represents the increase (with respect to averages) of the A_{1610}/A_{1571} ratio for the fiber with MWCNTs (TF-CNT, for instance) relative to its corresponding fiber without MWCNTs (TF, for instance). This parameter may be interpreted as an indicator of the amount of MWCNTs on the fiber surface. As shown in Table 3, the as-received fiber and the fiber treated with chlorosulfonic acid (TFCL) present higher amounts of MWCNTs deposited on their surface (ratios of 20.3 and 18.7, respectively). In the case of TF, the functional groups present in the oxidized MWCNTs (OH, C-O and COOH) interact with the oxygen-containing functional groups present in the FSC of the as-received fiber, as reported for glass and carbon fibers [19,38]. Twaron fibers have a standard surface coating which comprises a non-ionic emulsifier containing large molecules with ramifications of ethylene and propylene oxide functional groups, as well as hydroxyl and carboxyl groups [39,40]. Upon ultrasonic MWCNT deposition, the temperature is raised to $\sim 70^\circ\text{C}$ and the hydroxyl and carboxyl groups of the oxidized MWCNTs can covalently react with the ethylene and propylene oxides through anionic ring-opening reactions [41]. These ring-opening reactions yield new hydroxyl groups, which could also react with the hydroxyl and carboxyl groups on the MWCNT surface through hydrogen bonding [19].

On the other hand, according to the increase in the area ratio of 18.7%, the chlorosulfonic treatment also promotes good affinity between the treated fibers and the oxidized MWCNTs, being mild with the fiber. In contrast, the acid treatment TFAC is aggressive with the fiber and concomitant with the removal of the FSC renders the lowest amount of MWCNTs per area (9.6% increase).

3.2.2. X-ray photoelectron spectroscopy

Fig. 12 shows the XPS results of TF-CNT, TFCL-CNT and TFAC-CNT fibers. Fig. 12a shows surveys of the fibers with deposited MWCNTs. These spectra show high intensity bands corresponding to C1s, N1s and O1s orbitals; additionally, the spectra show a few low intensity bands corresponding to traces of metals, likely from the MWCNT synthesis [24,42].

A more detailed analysis is obtained by examining the deconvolution of the C1s band shown in Fig. 12b–d. In Fig. 12b five bands are observed corresponding to O=C–O (289 eV), O=C–NH/C=O (287.8 eV), C–O (286 eV), C–C (284.6 eV) [6] and C–H (283 eV) [43]. The first one from left to right, located $\sim 289\text{ eV}$ corresponds to the contribution of C=O and C–O bonds (7.4%), probably coming from the interaction between the functional groups of the oxidized MWCNTs and the FSC. This band was not observed in Figs. 12c–d (TFCL-CNT and TFAC-CNT), where the FSC was removed. The emergence of a new band at $\sim 289\text{ eV}$ in as-received fibers after MWCNT deposition suggests an interaction between the FSC and the MWCNTs, as has been previously reported [19,44]. The increase of the intensity of the band at $\sim 287.8\text{ eV}$ (O=C–NH/C=O) in Figs. 12b–d in comparison to the fibers without MWCNTs (Figs. 8b–d) suggests the presence of oxidized MWCNTs. The band at $\sim 286\text{ eV}$ corresponds to C=O bonds. Its contribution is higher for TFCL-CNT (22.5%, Fig. 12c) than for TF-CNT (20.3%, Fig. 12a) and even higher for TFAC-CNT (32.2%, Fig. 12d), suggesting an increased contribution of the C–O bonds coming from the oxidation of the aramid fibers and the oxidized MWCNTs.

3.2.3. Scanning electron microscopy

Fig. 13 shows SEM micrographs of the aramid fiber yarns with MWCNTs deposited on their surface at magnifications of 100x, 1000x and 5000x (left to right). Fig. 13a shows as-received fibers with deposited MWCNTs (TF-CNT), showing a relatively homogeneous distribution of MWCNTs over the TF surface; a few agglomerations are observed in some regions. Fig. 13b shows SEM micrographs of TFCL-CNT fibers, where again a rather homogeneous distribution of MWCNTs on the fiber surface with few agglomerates is observed. However, Fig. 13c shows that for TFAC-CNT fibers the MWCNT distribution is significantly less homogeneous than for TF-CNT and TFCL-CNT fibers. This is probably caused by the removal of the FSC and the presence of a high density of functional groups in localized areas of the fiber surface. These

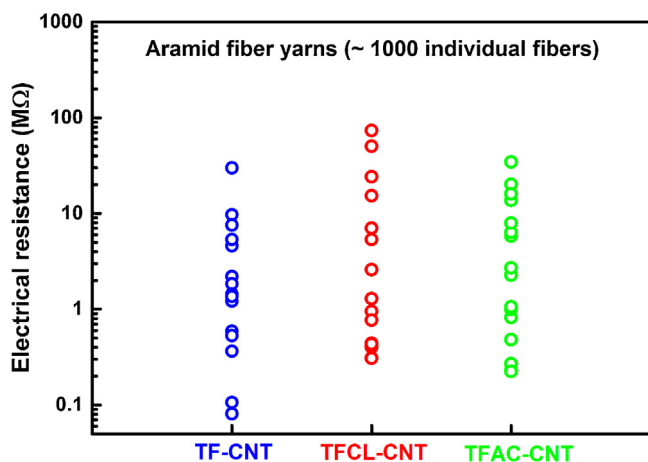


Fig. 15. Electrical resistance of TF-CNT, TFCL-CNT and TFAC-CNT fibers.

results are in agreement with the results of Raman spectroscopy and XPS.

3.2.4. Raman spectroscopy mapping

Fig. 14 shows Raman mapping of the aramid fiber surface containing the deposited MWCNTs. Bright regions represent areas of high intensity of the MWCNT 2D band, while dark regions are fiber areas with low intensity of such a band, i.e. low density of MWCNTs. Mapping the TF-CNT fiber surface (Fig. 14a) consistently shows many MWCNT-rich zones. The distribution of MWCNTs on the fiber surface looks relatively more homogenous for TFCL-CNT (Fig. 14b), given the mild chlorosulfonic acid treatment carried out. However, the map of the TFAC-CNT fibers (Fig. 14c) evidence larger MWCNT agglomerations, likely due to the removal of the FSC and the aggressiveness of the fiber treatment, leading to localized active regions. These results are again in agreement with the SEM and physico-chemical characterizations presented earlier.

3.2.5. Electrical resistance of MWCNT-modified fibers

Twaron fibers are electrically insulating materials with conductivity of the order of 1×10^{-15} S/cm [45]. Upon MWCNT deposition, the surface of the twaron fibers became electrically conductive. Fig. 15 shows the electrical resistances of 10 mm long fiber yarns for TF-CNT, TFCL-CNT and TFAC-CNT fibers. After the deposition of MWCNTs, the majority of the electrical resistance values of the three samples fall in the range of 200 k Ω –50 M Ω , and a few TF-CNT fibers reach \sim 100 k Ω . MWCNTs deposited onto the surface of aramid fibers promote the formation of electrically conductive pathways among adjacent fibers and along the yarn, and the chemical treatment does not seem to yield an important global effect in the surface conductivity of the fiber. The data scattering might seem high, but this experimental scattering is reasonable considering the challenges involved in the deposition of MWCNTs onto engineering fibers, due to the difficulty of homogeneously depositing MWCNTs at the nano- and micro-scales (see e.g. [20]). This electrical property confers multifunctional capabilities to the aramid fibers, such as the ability to be used as electrical filaments [46,47], as strain sensors [48] or for “structural health” monitoring in multiscale hierarchical composites [49,50].

4. Conclusions

Oxidized multiwall carbon nanotubes were deposited on as-received (commercial) Twaron fibers and on chemically modified Twaron fibers (with “sizing” removed) using chlorosulfonic acid at 0.2% w/w or using a mixture of nitric and sulfuric acids at 3.0 M.

FT-IR and Raman analyses showed certain evidence of the success of the chemical treatments, but the surface changes detected by these techniques were only moderate due to the high crystallinity of the aramid fiber. XPS, however, showed conspicuous evidence of fiber surface changes produced by both chemical treatments. O=C-NH/C=O and C-O functional groups were observed in higher proportion for fibers modified with the nitric and sulfuric acids mixture, in comparison to those generated for fibers treated with the chlorosulfonic acid or as-received. This suggests that the fiber modification based on the nitric and sulfuric acids mixture generates more functional groups, but is also more aggressive. Both chemical treatments modified the mechanical properties of the fibers. The reduction of strength, ultimate strain, and elastic modulus of the fibers treated with chlorosulfonic acid was small. However, those treated with the mixture of nitric and sulfuric acids experienced more important reductions in their mechanical properties.

Regarding MWCNT deposition on the surface of aramid fibers, Raman spectroscopy and XPS presented conspicuous evidence of the MWCNT deposition on the aramid fiber. The results suggest that the presence of the fiber surface coating or a mild chlorosulfonic treatment promote increased interactions between the fiber and oxidized MWCNTs. Confirmation of these results was obtained by SEM images and Raman intensity maps, pointing towards a more homogeneous distribution of MWCNTs on the fiber surface for fibers without treatment and for those treated with chlorosulfonic acid.

The results of electrical resistance measurements of 10 mm long fiber yarns span over similar values (0.2–50 M Ω) for as-received or chemically modified fibers. The similarity of these values regardless of the fiber treatment suggests that the establishment of electrically percolative networks occurs predominantly by side-contact between adjacent fibers, and is not greatly affected by the fiber treatment. The electrical resistance of the fiber yarns (\sim 200 k Ω /cm) and their excellent mechanical properties, render multifunctional properties that may be exploited in applications related to energy storage and transportation, as well as structural elements in multiscale hierarchical composites which are able to self-sense their damage.

Acknowledgements

This research is part of a Mexico-Chile collaborative project supported by CONACYT-CIAM (Mexico, No. 188089) and CONICYT (Chile, No. 120003) directed by Dr. Avilés and Dr. Yazdani-Pedram. Input of Dr. Juan V. Cauich regarding the chemical treatments is strongly appreciated. Support in the form of a grant to our graduate school program from the “Fondo Mixto CONACYT-Gobierno del Estado de Yucatán”, project No. 247046, is also acknowledged. Technical assistance of Cesar Martin and Alejandro May (CICY) is also strongly appreciated.

References

- [1] H.G. Chae, S. Kumar, Rigid-rod polymeric fibers, *J. Appl. Polym. Sci.* 100 (2006) 791–802.
- [2] D. Tanner, J.A. Fitzgerald, B.R. Phillips, The kevlar story—an advanced materials case study, *Angewandte Chemie International Edition in English* 28 (1989) 649–654.
- [3] W. Chen, X.M. Qian, X.Q. He, Z.Y. Liu, J.P. Liu, Surface modification of Kevlar by grafting carbon nanotubes, *J. Appl. Polym. Sci.* 123 (2012) 1983–1990.
- [4] T. Lin, S. Wu, J. Lai, S. Shyu, The effect of chemical treatment on reinforcement/matrix interaction in Kevlar-fiber/bismaleimide composites, *Compos. Sci. Technol.* 60 (2000) 1873–1878.
- [5] M. Jassal, S. Ghosh, Aramid fibres—An overview, *Indian J. Fibre Text.* 27 (2002) 290–306.
- [6] C. Jia, P. Chen, B. Li, Q. Wang, C. Lu, Q. Yu, Effects of Twaron fiber surface treatment by air dielectric barrier discharge plasma on the interfacial adhesion in fiber reinforced composites, *Surf. Coat. Technol.* 204 (2010) 3668–3675.

- [7] C.X. Wang, M. Du, J.C. Lv, Q.Q. Zhou, Y. Ren, G.L. Liu, D.W. Gao, L.M. Jin, Surface modification of aramid fiber by plasma induced vapor phase graft polymerization of acrylic acid. I. Influence of plasma conditions, *Applied Surface Science* 349 (2015) 333–342.
- [8] J. Chen, Y. Zhu, Q. Ni, Y. Fu, X. Fu, Surface modification and characterization of aramid fibers with hybrid coating, *Appl. Surf. Sci.* 321 (2014) 103–108.
- [9] S. Hussain, C. Yorucu, I. Ahmed, R. Hussain, B. Chen, M. Bilal Khan, N.A. Siddique, I.U. Rehman, Surface modification of aramid fibres by graphene oxide nano-sheets for multiscale polymer composites, in: *Surf. Coat. Technol.* (2014) 458–466.
- [10] M. Xi, Y.-L. Li, S.-y. Shang, D.-H. Li, Y.-X. Yin, X.-Y. Dai, Surface modification of aramid fiber by air DBD plasma at atmospheric pressure with continuous on-line processing, *Surf. Coat. Technol.* 202 (2008) 6029–6033.
- [11] T. Ai, R. Wang, W. Zhou, Effect of grafting alkoxy silane on the surface properties of Kevlar fiber, *Polym. Compos.* 28 (2007) 412–416.
- [12] J. Gao, Y. Dai, X. Wang, J. Huang, J. Yao, J. Yang, X. Liu, Effects of different fluorination routes on aramid fiber surface structures and interlaminar shear strength of its composites, *Appl. Surf. Sci.* 270 (2013) 627–633.
- [13] S. Wu, G. Sheu, S. Shyu, Kevlar fiber–epoxy adhesion and its effect on composite mechanical and fracture properties by plasma and chemical treatment, *J. Appl. Polym. Sci.* 62 (1996) 1347–1360.
- [14] J. Maity, C. Jacob, C. Das, A. Kharitonov, R. Singh, S. Alam, Fluorinated aramid fiber reinforced polypropylene composites and their characterization, *Polym. Compos.* 28 (2007) 462–469.
- [15] J. Guo, C. Lu, Continuous preparation of multiscale reinforcement by electrophoretic deposition of carbon nanotubes onto carbon fiber tows, *Carbon* 50 (2012) 3101–3103.
- [16] S. Zhang, W. Liu, L. Hao, W. Jiao, F. Yang, R. Wang, Preparation of carbon nanotube/carbon fiber hybrid fiber by combining electrophoretic deposition and sizing process for enhancing interfacial strength in carbon fiber composites, *Compos. Sci. Technol.* 88 (2013) 120–125.
- [17] L. Liu, P.-C. Ma, M. Xu, S.U. Khan, J.-K. Kim, Strain-sensitive Raman spectroscopy and electrical resistance of carbon nanotube-coated glass fibre sensors, *Compos. Sci. Technol.* 72 (2012) 1548–1555.
- [18] J. Zhang, R. Zhuang, J. Liu, E. Mäder, G. Heinrich, S. Gao, Functional interphases with multi-walled carbon nanotubes in glass fibre/epoxy composites, *Carbon* 48 (2010) 2273–2281.
- [19] J. Ku-Herrera, F. Avilés, A. Nistal, J. Cauch-Rodríguez, F. Rubio, J. Rubio, P. Bartolo-Pérez, Interactions between the glass fiber coating and oxidized carbon nanotubes, *Appl. Surf. Sci.* 330 (2015) 383–392.
- [20] J. Rausch, E. Mäder, Health monitoring in continuous glass fibre reinforced thermoplastics: manufacturing and application of interphase sensors based on carbon nanotubes, *Compos. Sci. Technol.* 70 (2010) 1589–1596.
- [21] N.A. Siddiqui, M.-L. Sham, B.Z. Tang, A. Munir, J.-K. Kim, Tensile strength of glass fibres with carbon nanotube–epoxy nanocomposite coating, *Composites Part A: Applied Science and Manufacturing* 40 (2009) 1606–1614.
- [22] I. O'Connor, H. Hayden, J.N. Coleman, Y.K. Gun'ko, High-Strength, high-Toughness composite fibers by swelling kevlar in nanotube suspensions, *Small* 5 (2009) 466–469.
- [23] G. Derombise, L. Vouyovitch Van Schoors, M.F. Messou, P. Davies, Influence of finish treatment on the durability of aramid fibers aged under an alkaline environment, *J. Appl. Polym. Sci.* 117 (2010) 888–898.
- [24] F. Avilés, A. May-Pat, G. Canché-Escamilla, O. Rodríguez-Uicab, J.J. Ku-Herrera, S. Duarte-Aranda, J. Uribe-Calderon, P.I. Gonzalez-Chi, L. Arronche, V. La Saponara, Influence of carbon nanotube on the piezoresistive behavior of multiwall carbon nanotube/polymer composites, *J. Intell. Mater. Syst. Struct.* 17 (2014) 1–12.
- [25] F. Avilés, J.V. Cauch-Rodríguez, L. Moo-Tah, A. May-Pat, R. Vargas-Coronado, Evaluation of mild acid oxidation treatments for MWCNT functionalization, *Carbon* 47 (2009) 2970–2975.
- [26] E. Chatzi, J. Koenig, Morphology and structure of Kevlar fibers: a review, *Polymer-Plastics Technology and Engineering* 26 (1987) 229–270.
- [27] R. Uppal, G.N. Ramaswamy, T. Loughin, A novel method to assess degree of crystallinity of aramid filament yarns, *J. Ind. Text.* 41 (2012) 3–19.
- [28] I. O'Connor, H. Hayden, S. O'Connor, J.N. Coleman, Y.K. Gun'ko, Kevlar coated carbon nanotubes for reinforcement of polyvinylchloride, *J. Mater. Chem.* 18 (2008) 5585–5588.
- [29] S. Hofmann, Practice of surface and interface analysis with AES and XPS, in: *Auger- and X-ray Photoelectron Spectroscopy in Materials Science*, Springer Berlin Heidelberg, 2013, pp. 409–449.
- [30] S. Tougaard, Quantification of surface and near-surface composition by AES and XPS, in: *Handbook of Surface and Interface Analysis—Methods for Problem-Solving*, CRC Press Taylor & Francis Group, Boca Raton, 2009, pp. 223–244.
- [31] M.S. Dresselhaus, G. Dresselhaus, R. Saito, A. Jorio, Raman spectroscopy of carbon nanotubes, *Physics reports* 409 (2005) 47–99.
- [32] D.F. Adams, L.A. Carlsson, R.B. Pipes, *Experimental Characterization of Advanced Composite Materials*, CRC Press, Boca Raton, Florida, 2003.
- [33] J. Maity, C. Jacob, C. Das, S. Alam, R. Singh, Direct fluorination of Twaron fiber and the mechanical, thermal and crystallization behaviour of short Twaron fiber reinforced polypropylene composites, *Composites Part A: Applied Science and Manufacturing* 39 (2008) 825–833.
- [34] J. Zhao, Effect of surface treatment on the structure and properties of para-aramid fibers by phosphoric acid, *Fibers Polym.* 14 (2013) 59–64.
- [35] T. Lin, B. Kuo, S. Shyu, S.-H. Hsiao, Improvement of the adhesion of Kevlar fiber to bismaleimide resin by surface chemical modification, *J. Adhes. Sci. Technol.* 13 (1999) 545–560.
- [36] L. Penn, F. Milanovich, Raman spectroscopy of kevlar 49 fibre, *Polymer* 20 (1979) 31–36.
- [37] L.E. Hamilton, P. Sherwood, B.M. Reagan, X-ray photoelectron spectroscopy studies of photochemical changes in high-performance fibers, *Appl. Spectrosc.* 47 (1993) 139–149.
- [38] L. Liu, Q. Jiang, T. Zhu, X. Guo, Y. Sun, Y. Guan, Y. Qiu, Influence of moisture regain of aramid fibers on effects of atmospheric pressure plasma treatment on improving adhesion with epoxy, *J. Appl. Polym. Sci.* 102 (2006) 242–247.
- [39] P. De Lange, P. Akker, A. Maas, A. Knoester, H. Brongersma, Adhesion activation of Twaron® aramid fibres studied with low-energy ion scattering and x-ray photoelectron spectroscopy, *Surf. Interface Anal.* 31 (2001) 1079–1084.
- [40] P.J. De Lange, E. Mäder, K. Mai, R.J. Young, I. Ahmad, Characterization and micromechanical testing of the interphase of aramid-reinforced epoxy composites, *Composites Part A: Applied Science and Manufacturing* 32 (2001) 331–342.
- [41] M.S. Thompson, T.P. Vadala, M.L. Vadala, Y. Lin, J.S. Riffle, Synthesis and applications of heterobifunctional poly(ethylene oxide) oligomers, *Polymer* 49 (2008) 345–373.
- [42] V.N. Popov, Carbon nanotubes: properties and applications, *Mater. Sci. Technol.* 43 (2004) 61–102.
- [43] S.-H. Zhang, G.-Q. He, G.-Z. Liang, H. Cui, W. Zhang, B. Wang, Comparison of F-12 aramid fiber with domestic aramid fiber III on surface feature, *Appl. Surf. Sci.* 256 (2010) 2104–2109.
- [44] M. Li, Y. Gu, Y. Liu, Y. Li, Z. Zhang, Interfacial improvement of carbon fiber/epoxy composites using a simple process for depositing commercially functionalized carbon nanotubes on the fibers, *Carbon* 52 (2013) 109–121.
- [45] K.K. Chang, Aramid fibers, in: *Materials Park*, ASM International, OH, 2001, pp. 41–45.
- [46] N. Behabtu, C.C. Young, D.E. Tsentelovich, O. Kleinerman, X. Wang, A.W. Ma, E.A. Bengio, R.F. ter Waarbeek, J.J. de Jong, R.E. Hoogerwerf, Strong light, multifunctional fibers of carbon nanotubes with ultrahigh conductivity, *Science* 339 (2013) 182–186.
- [47] C. Xiang, W. Lu, Y. Zhu, Z. Sun, Z. Yan, C.-C. Hwang, J.M. Tour, Carbon nanotube and graphene nanoribbon-Coated conductive kevlar fibers, *ACS Appl. Mater. Interfaces* 4 (2011) 131–136.
- [48] G.J. Ehlert, H.A. Sodano, Fiber strain sensors from carbon nanotubes self-assembled on aramid fibers, *J. Intell. Mater. Syst. Struct.* (2014) 1–5.
- [49] D.D. Chung, Self-monitoring structural materials, *Materials Science and Engineering: R: Reports* 22 (1998) 57–78.
- [50] R. Salvado, C. Lopes, L. Szojda, P. Araújo, M. Gorski, F.J. Velez, J. Castro-Gomes, R. Krzywon, Carbon fiber epoxy composites for both strengthening and health monitoring of structures, *Sensors* 15 (2015) 10753–10770.